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# Highly efficient crystalline-amorphous Fe<sub>2</sub>O<sub>3</sub>/Fe-OOH oxygen evolution electrocatalysts reconstructed by FeS<sub>2</sub> nanoparticles

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## ABSTRACT

It is crucial to explore the earth-abundant, high-efficient and stable electrocatalysts suitable for accelerating water splitting kinetics of oxygen evolution reaction (OER) in alkaline electrolytes. Herein, Fe<sub>x</sub>Co<sub>1-x</sub>S<sub>2</sub> (x = 0, 0.25, 0.5, 0.75 and 1) nanoparticles were fabricated via hydrothermal method and sulfidizing. Remarkably, the FeS<sub>2</sub> delivers the most outstanding OER activity among prepared catalysts, with the overpotential of 240 mV for 10 mA/cm<sup>2</sup>, ultra-low Tafel slope of 45 mV/decades and high stability for 16 h. XRD, SEM, HRTEM, Raman and XPS characteristics before and OER process, TOF and electrochemical results indicate that surface reconstruction, strong combination of Fe-S and uniform dispersion of particles endow FeS<sub>2</sub> with excellent catalytic property. The strong combination of Fe-S bond and uniform dispersion of FeS<sub>2</sub> facilitates electron transfer. And the reconstructed crystalline-amorphous Fe<sub>2</sub>O<sub>3</sub>/FeOOH with rough surface may have a low energy barrier for the elementary reaction of O<sup>\*</sup>→OOH<sup>\*</sup>, which is responsible for boosting the generation of O<sub>2</sub>. In the practical two-electrode overall water splitting system in 1 M KOH, the FeS<sub>2</sub>/NF//Pt/C/NF system only needs an ultra-low cell voltage of 1.46 and 1.642 V to generate the current densities of 10 and 50 mA/cm<sup>2</sup>, respectively.

## 1. Introduction

Hydrogen is widely regarded as a key component in future energy systems just as it is a clean, sustainable and transportable energy carrier (Pomerantseva et al., 2019; Jiao et al., 2015; Zhang et al., 2023a; Zhang et al., 2023b). Electrocatalytic water splitting, which can driven by electricity of power grid, solar energy and rechargeable batteries (Rebrov and Gao, 2023; Kang et al., 2021) is extensively considered an environmentally friendly and efficient hydrogen production technology (Suen et al., 2017; Gong et al., 2013; He et al., 2024). Compared with the hydrogen evolution reaction (HER), the electrocatalytic efficiency is severely restricted by the oxygen evolution reaction (OER) process because of the slow kinetics and high overpotentials resulting from the complex four-electron transfer (Zhang et al., 2023a; Zhang et al., 2023b; Wang et al., 2020d; Wang et al., 2023d). So, there is a pressing need to explore highly efficient, low-cost and earth-abundant electrocatalysts with fast kinetics and low overpotential for water splitting. Despite noble metal catalysts have proven to be desirable catalysts for OER, the high cost and scarcity of resources seriously limited their wide applications (Ying et al., 2018; Su et al., 2024).

So far, various non-precious materials, including metal oxides (Zhang et al., 2020), phosphate (Li et al., 2022), chalcogenides (Yue et al., 2023), and hydroxides (Peng et al., 2023) have been designed to catalyze the OER, in which Cobalt (Co) or iron (Fe)-based chalcogenides, such as CoFe-Co<sub>8</sub>FeS<sub>8</sub> (Wang et al., 2019), Co<sub>8</sub>FeS<sub>8</sub>/CoS (Wang et al., 2020a), are excellent candidates towards OER because of their good electronic conductivity and their rich redox properties. Among them, MS<sub>2</sub> (M=Fe, Co) deliver high electrocatalytic activity towards OER (Li et al., 2018; Wang et al., 2023c; Xie et al., 2022). Li et al prepared Fe<sub>7</sub>S<sub>8</sub>/FeS<sub>2</sub>/C (Xu et al., 2021) electrocatalysts, in which only 262 mV of overpotentials was required to reach 10 mA/cm<sup>2</sup> towards OER. The synthesized FeCoS<sub>2</sub>/Co<sub>4</sub>S<sub>3</sub>/NFG catalysts exhibited enhanced OER catalytic activity, with an overpotential of 276 mV for OER at 10 mA/cm<sup>2</sup> (Wang et al., 2023c). Until now, we should also do much research on preparing MS<sub>2</sub> (M=Fe, Co) catalysts with improved electrocatalytic performance and reduced energy consumption.

Exploring the real active site during the OER process in alkaline environments is very important for improving the catalytic performance. Recently, a large number of researchers has put forward a view that the prepared electrocatalysts are not real “catalysts”, but “pre-

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catalysts”, for which their morphology, composition and phase changed dramatically (Chen et al., 2018; Wang et al., 2018; Ren et al., 2023). Huang et al (Su et al., 2024) found that the prepared  $\text{RuO}_2/\text{Co}_3\text{O}_4$  had reconstructed to  $\text{RuO}_2/\text{CoOOH}$  during the OER process, which showed a low energy barrier for the reaction of  $\text{O}^*$  to  $^*\text{OOH}$  to improve OER performance. Liu et al (Yue et al., 2023) proved that  $\text{FeS}_2@/\text{FeOOH}$  reconstructed by  $\text{FeS}_2$  microspheres during the OER process was responsible for the improved OER activity. Also, the high entropy sulfide of  $\text{FeNiCoCrMnS}_2$  was reconstructed to the metal (oxy)hydroxide during the OER process (Nguyen et al., 2021). Whereas, studying the synergistic effect of Co contents and reconstructing influence on Fe/S catalysts is still rare.

In this study,  $\text{Fe}_x\text{Co}_{1-x}\text{S}_2$  ( $x = 0, 0.25, 0.5, 0.75$  and  $1$ ) nanoparticles supported on Ni foam were designed, which exhibit excellent OER activity in alkaline environments. The electrochemical measurements of LSV, CV, EIS, i-t for  $\text{Fe}_x\text{Co}_{1-x}\text{S}_2$  catalysts were performed to evaluate the electrocatalytic activity for OER. Remarkably, the  $\text{FeS}_2$  delivers the most outstanding OER activity among prepared  $\text{Fe}_x\text{Co}_{1-x}\text{S}_2$  catalysts. The strong combination of Fe-S bond and uniform dispersion endow  $\text{FeS}_2$  fast electron transfer. By comparing with XRD, XPS, SEM, TEM and HRTEM, we found that the crystalline-amorphous  $\text{Fe}_2\text{O}_3/\text{Fe}-\text{OOH}$  with the rough surface after OER reconstructed by  $\text{FeS}_2$  supported on Ni foam is the real active site. The reconstructed crystalline-amorphous  $\text{Fe}_2\text{O}_3/\text{Fe}-\text{OOH}$  may have a low energy barrier for the elementary reaction of  $\text{O}^* \rightarrow \text{OOH}^*$ . Moreover, it only requires the voltage of 1.460 and 1.642 V to generate the current density of 50 and 100  $\text{mV}/\text{cm}^2$  in a practical two-electrode overall water splitting system. This study may provide a new idea for the research of metal sulfide electrocatalysts toward OER for alkaline water splitting systems.

## 2. Experimental section

### 2.1. Samples preparation

As shown in Scheme 1, for preparing  $\text{CoS}_2$  nanoparticles: 0.6 g Lysine was added to 75 mL  $\text{CoCl}_2$  solution (0.05 mol/L) and stirred at  $40^\circ\text{C}$  for 0.5 h in a water bath. 4 mL of ammonia water (32 wt%) was evenly added to the solution within 0.5 h using a peristaltic pump. After stirring the solution for 0.5 h, transfer it into a 100 mL Teflon-lined autoclave and heat it in an oven at  $180^\circ\text{C}$  for 6 h. After centrifugal, washing and drying for three times, the obtained powder was vacuum-dried overnight at  $60^\circ\text{C}$ . The dried sample was thoroughly mixed with sublimed sulfur at a mass ratio of 1:5, and ground for 10 min. The final  $\text{CoS}_2$  sample was obtained by calcining the mixed powder at  $450^\circ\text{C}$  in  $\text{N}_2$  atmosphere for 4 h, with a heating rate of  $5^\circ\text{C}/\text{min}$ .

For preparing  $\text{Fe}_x\text{Co}_{1-x}\text{S}_2$  nanoparticles: the synthesis process is

similar to the above except for changing the amounts of the added salts. For  $x = 0, 0.25, 0.5, 0.75$  and  $1$ , the amounts of added  $\text{CoCl}_2$  are 75 mL, 56.25 mL, 37.5 mL, 18.75 mL, and 0 mL while the amounts of added  $\text{FeCl}_3$  are 0 mL, 18.75 mL, 37.5 mL, 56.25 mL, and 75 mL, and the samples are marked with  $\text{CoS}_2$ ,  $\text{Fe}_{0.25}\text{Co}_{0.75}\text{S}_2$ ,  $\text{Fe}_{0.5}\text{Co}_{0.5}\text{S}_2$ ,  $\text{Fe}_{0.75}\text{Co}_{0.25}\text{S}_2$ , and  $\text{FeS}_2$ , respectively.

### 2.2. Material characterizations

The surface structure, morphology and interior structure of the synthetic samples were analyzed by emission scanning electron microscope (FESEM, Zeiss Sigma, Japan) equipped with energy-dispersive X-ray spectroscopy (EDS) and high-resolution transmission electron microscope (HRTEM, FEI Tecnai F30, America). The crystal structures of the samples were investigated using X-ray diffraction (XRD, Smartlab, Rigaku, Japan) with a  $\text{CuK}\alpha$  radiation source. The elementary composites and surface chemical status of the samples were inspected by performing X-ray photoelectron spectroscopy (XPS, ESCALAB 250xi) and Raman spectroscopy (Raman, Thermo DXRxi).

### 2.3. Electrochemical measurements

All the electrochemical measurements were performed by using an electrochemical workstation of Gamry, equipped with a conventional three-electrode cell. In the three-electrode system, Ni foam ( $1 \times 1 \text{ cm}^2$ ) coated with our samples, graphite rod, and  $\text{Hg}/\text{HgO}$  served as working electrode, counter electrode, and reference electrode, respectively. The working electrode was prepared as follow: 10 mg catalysts and 1.0 mg conductive carbon were dispersed onto a mixture of 400  $\mu\text{L}$  isopropanol and 80  $\mu\text{L}$  Nafion solution (5 %, DuPont) with ultrasound for 90 min. After that the prepared homogeneous ink was uniformly dropped onto the pre-treated foam Ni. The conducted potentials in this measuring system were converted to the reversible hydrogen electrode (RHE) potential using the following equation:

$$E_{\text{RHE}} = E_{\text{Hg}/\text{HgO}} + 0.059 \text{ pH} + 0.098$$

LSV cures were conducted at a scan rate of 5  $\text{mV}/\text{s}$  in the range of 0–1 V vs.  $\text{Hg}/\text{HgO}$  with 95 % iR-corrected. Cyclic voltammetry (CV) scanning was carried out in 1 M KOH and was used to inspect the electrochemical double-layer capacitance ( $C_{\text{dl}}$ ) and the roughness factor ( $R_f$ ) of the samples according to the equation:  $R_f = C_{\text{dl}}/60$ , where 60 represents the assumed 60  $\mu\text{F}/\text{cm}^2$  for the double layer capacitance of a smooth surface (Zhang et al., 2013).

Electrochemical impedance spectroscopy (EIS) was conducted to collect the Nyquist plots towards OER. To evaluate the OER stability, a



Scheme 1. The preparing procedure of the  $\text{Fe}_x\text{Co}_{1-x}\text{S}_2$  nanoparticles.

chronopotentiometry test was performed for 16 h at a potential of 1.547 vs. RHE.

Also, a two-electrode system was set up for practical overall water splitting. All the two- and three-electrode systems are alkaline solutions with 1.0 M KOH.

### 3. Results and discussion

#### 3.1. Characterization and electrocatalytic performance for OER

The surface structure and morphologies of  $\text{CoS}_2$ ,  $\text{Fe}_{0.25}\text{Co}_{0.75}\text{S}_2$ ,  $\text{Fe}_{0.5}\text{Co}_{0.5}\text{S}_2$ ,  $\text{Fe}_{0.75}\text{Co}_{0.25}\text{S}_2$  and  $\text{FeS}_2$  were investigated by the FESEM, as shown in Fig. 1. All the  $\text{Fe}_x\text{Co}_{1-x}\text{S}_2$  ( $x = 0, 0.25, 0.5, 0.75, 1$ ) samples are composed of nanoparticles. In Fig. 1a, the  $\text{CoS}_2$  is made up of very fine nanoparticles. Interestingly, by increasing the  $x$  value from 0 to 1 for  $\text{Fe}_x\text{Co}_{1-x}\text{S}_2$ , the sample became more bigger and uniform, with more smoother surfaces. The uniform particle dispersion is beneficial to electron transport, thereby improving electrochemical performance and reducing resistance for OER (Wang et al., 2022). The elemental mapping of  $\text{FeS}_2$  in Fig. 1f indicates that the dispersion of S and Fe elements is consistent.

TEM and HRTEM were carried out to inspect the interior structure of the synthetic  $\text{FeS}_2$  sample, as displayed in Fig. 2. Fig. 2a-2b show that the  $\text{FeS}_2$  nanoparticles are uniformly dispersed with the size of 50 nm, which is consistent with the results in Fig. 1e-1f. The HRTEM images in Fig. 2c-2f clearly show that the particles have a clear and smooth surface, which will under significant changes during the OER process. Additionally, the determined lattice fringes are 0.27, 0.31, and 0.19 nm, corresponding to the (200), (111), and (220) planes of  $\text{FeS}_2$  (PDF#71-1680), respectively.

XRD patterns of various samples are shown in Fig. 3. In the patterns of  $\text{CoS}_2$ , the diffraction peaks at  $27.878^\circ$ ,  $32.300^\circ$ ,  $36.237^\circ$ ,  $39.835^\circ$ ,  $46.329^\circ$  and  $54.938^\circ$  are attributed to (111), (200), (210), (211), (220), and (311) planes of  $\text{CoS}_2$  (PDF#89-1492). With increasing the amounts of Fe, all the diffraction peaks shifted towards the large  $2\theta$  for the  $\text{Fe}_{0.5}\text{Co}_{0.5}\text{S}_2$ ,  $\text{Fe}_{0.75}\text{Co}_{0.25}\text{S}_2$  and  $\text{FeS}_2$ . This phenomenon is attributed to that the radius of Fe is smaller than that of Co. In the patterns  $\text{FeS}_2$ , the (111), (200), (210), (211), (220), and (311) planes match well with the  $2\theta$  of  $28.516^\circ$ ,  $33.045^\circ$ ,  $37.079^\circ$ ,  $40.768^\circ$ ,  $47.431^\circ$ ,  $56.278^\circ$  and belong to  $\text{FeS}_2$  (PDF#71-1680). The XRD patterns of the samples

indicate that we have obtained the pure  $\text{CoS}_2$ ,  $\text{Fe}_{0.5}\text{Co}_{0.5}\text{S}_2$ ,  $\text{Fe}_{0.75}\text{Co}_{0.25}\text{S}_2$ , and  $\text{FeS}_2$ .

XPS was performed to inspect the elementary composites and surface chemical status of the samples. The XPS full spectra in Fig. S1a show the obvious Co 2p, Fe 2p and S 2p peaks. The binding energies at 708.0 and 720.8 eV in Fig. S1b are assigned to  $\text{Fe}_{(\text{II})}\text{-S } 2p_{3/2}$  and  $\text{Fe}_{(\text{II})}\text{-S } 2p_{1/2}$ , respectively, while the peaks at 711.2 and 724 eV consist to  $\text{Fe}_{(\text{III})}\text{-S } 2p_{3/2}$  and  $\text{Fe}_{(\text{III})}\text{-S } 2p_{1/2}$  (Guo et al., 2020). Similarly, the XPS spectrum of Co 2p in Fig. S1c all can be decomposed into four peaks, where the binding energies at 778.9 and 798.7 eV belong to  $\text{Co}_{(\text{II})}\text{-S } 2p_{3/2}$  and  $\text{Co}_{(\text{II})}\text{-S } 2p_{1/2}$ . The other peaks at 782.9 and 803.8 eV are attributed to  $\text{Co}_{(\text{III})}\text{-S } 2p_{3/2}$  and  $\text{Co}_{(\text{III})}\text{-S } 2p_{1/2}$ , respectively (Fang et al., 2017; Yan et al., 2020). The two peaks at 163.2 and 164.5 eV for the S spectrum in Fig. S1d are assigned to S  $2p_{3/2}$  and S  $2p_{1/2}$  from metal sulfide. In addition, the peak at 169.3 eV may be due to the high oxidized sulfur species on the surface of  $\text{Fe}_{0.25}\text{Co}_{0.75}\text{S}_2$  samples (Khani and Wipf, 2017). The Raman spectra for  $\text{CoS}_2$ ,  $\text{Co}_{0.25}\text{Fe}_{0.75}\text{S}_2$  and  $\text{FeS}_2$  in Fig. S2 show that the peaks at around  $386$  to  $368 \text{ cm}^{-1}$  correspond to Co (Fe)-S bond. Interestingly, the peaks shifted towards lower  $\text{cm}^{-1}$ , which means the the binding energy became stronger by  $x$  increasing for  $\text{Fe}_x\text{Co}_{1-x}\text{S}_2$ , further effecting their ability for electrons transfer.

The LSV polarization curve was conducted to probe the OER performance of the various electrocatalysts, as displayed in Fig. 4a, and the corresponding overpotentials at 10, 50, and  $100 \text{ mA/cm}^2$  were shown in Fig. 4b. The overpotentials of  $\text{CoS}_2$ ,  $\text{Fe}_{0.25}\text{Co}_{0.75}\text{S}_2$ ,  $\text{Fe}_{0.5}\text{Co}_{0.5}\text{S}_2$ ,  $\text{Fe}_{0.75}\text{Co}_{0.25}\text{S}_2$ , and  $\text{FeS}_2$  at 10, 50, and  $100 \text{ mA/cm}^2$  are all smaller than that of Ni foam and benchmarking  $\text{RuO}_2$ , indicating the prepared  $\text{Fe}_x\text{Co}_{1-x}\text{S}_2$  ( $x = 0, 0.25, 0.5, 0.75, 1$ ) samples could act as an excellent catalyst for OER reaction under alkaline condition. Remarkably, the  $\text{FeS}_2$  delivers the most outstanding OER activity, and it just needs a low overpotentials of 240, 280, and 300 mV to gain the current densities of 10, 50, and  $100 \text{ mA/cm}^2$ , respectively. It is worth noting that the improved OER performance from  $\text{CoS}_2$  to  $\text{FeS}_2$  by increasing the  $x$  in  $\text{Fe}_x\text{Co}_{1-x}\text{S}_2$  may be attributed to the uniform dispersion, the clear and smooth surface and the inherent properties of Fe and Co bimetals, which will be beneficial to electron transport, consistent with the result in Figs. 1-2. On the other hand, the turnover frequency (TOF), acting as the intrinsic activity for the electrocatalysts during the OER process, was calculated with the overpotential of 300 mV and was shown in Supporting Information. The order of the oxygen binding sites for  $\text{FeS}_2$ ,

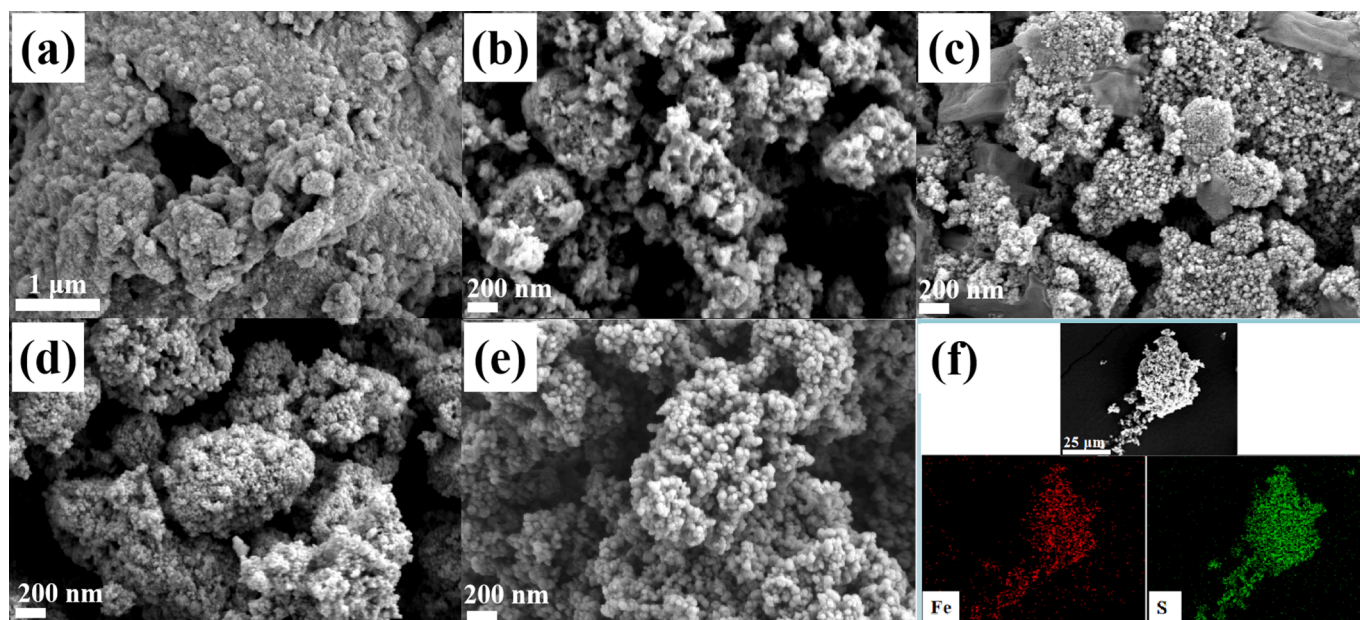


Fig. 1. SEM images of (a)  $\text{CoS}_2$  (b)  $\text{Fe}_{0.25}\text{Co}_{0.75}\text{S}_2$  (c)  $\text{Fe}_{0.5}\text{Co}_{0.5}\text{S}_2$  (d)  $\text{Fe}_{0.75}\text{Co}_{0.25}\text{S}_2$  (e)  $\text{FeS}_2$  and (f) elemental mapping of  $\text{FeS}_2$ .

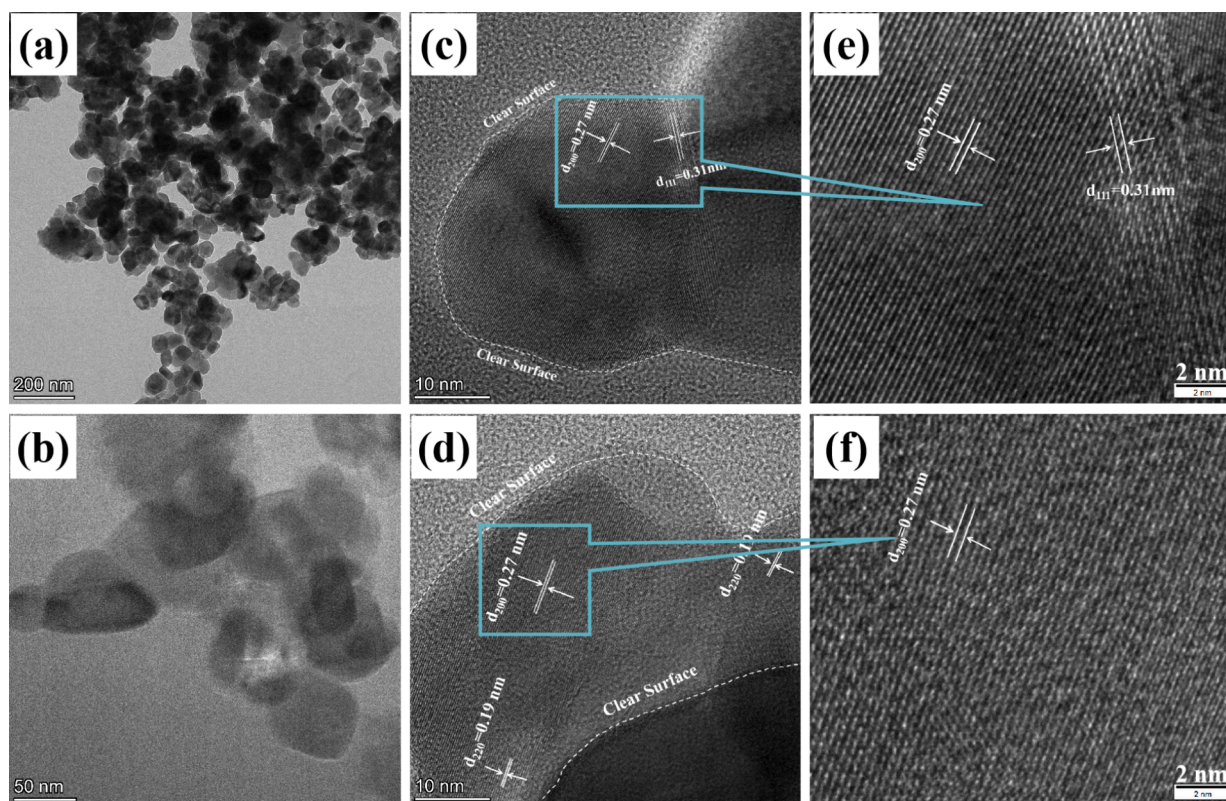


Fig. 2. TEM and HRTEM of FeS<sub>2</sub> before OER.

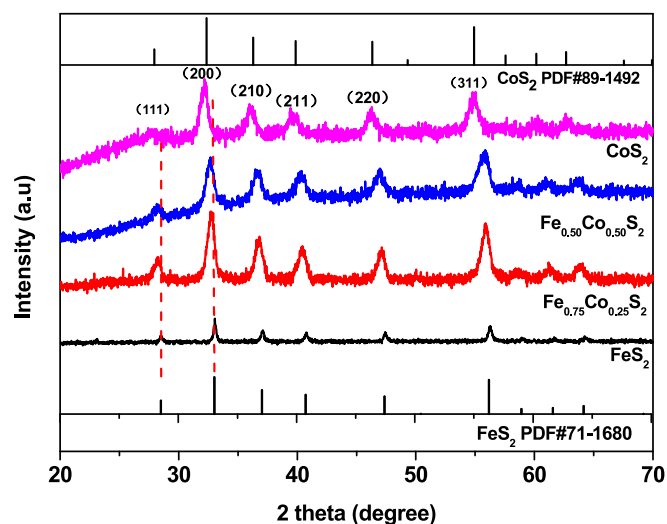


Fig. 3. XRD patterns of prepared samples.

Fe<sub>0.75</sub>Co<sub>0.25</sub>S<sub>2</sub>, Fe<sub>0.5</sub>Co<sub>0.5</sub>S<sub>2</sub>, Fe<sub>0.25</sub>Co<sub>0.75</sub>S<sub>2</sub> and CoS<sub>2</sub> are 5.01983 > 4.98769 > 4.95594 > 4.92488 > 4.89366, indicating that FeS<sub>2</sub> has the most strongest binding ability among the Fe<sub>x</sub>Co<sub>1-x</sub>S<sub>2</sub> samples. So, the TOFs of FeS<sub>2</sub>, Fe<sub>0.75</sub>Co<sub>0.25</sub>S<sub>2</sub>, Fe<sub>0.5</sub>Co<sub>0.5</sub>S<sub>2</sub>, Fe<sub>0.25</sub>Co<sub>0.75</sub>S<sub>2</sub>, CoS<sub>2</sub>, and RuO<sub>2</sub> are 0.03108, 0.01254, 0.00701, 0.00668, 0.00155, and 0.00154 O<sub>2</sub> per s per site, respectively. Astonishingly, the TOF of FeS<sub>2</sub> is 20 times more than that of CoS<sub>2</sub>, confirming the highest intrinsic activity for FeS<sub>2</sub>.

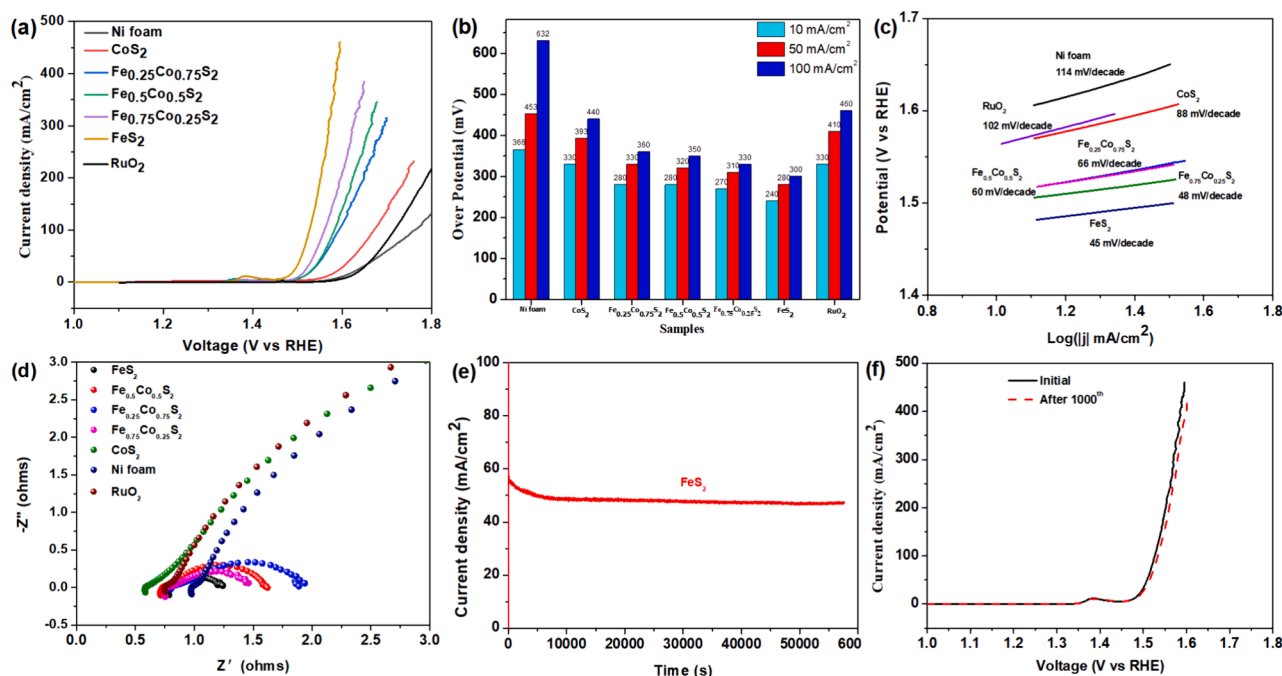
Tafel plots, reflecting the reaction kinetic of OER, were calculated from LSV data, as shown in Fig. 4c. The Tafel slopes of Fe<sub>x</sub>Co<sub>1-x</sub>S<sub>2</sub> catalysts are smaller than that of Ni foam and RuO<sub>2</sub>, indicating that they are easier for overcoming the reaction kinetic process than commercial RuO<sub>2</sub>. The smallest Tafel slope of 45 mV/decade for FeS<sub>2</sub> means it needs

overcome the smallest energy barrier during the Fe<sub>x</sub>Co<sub>1-x</sub>S<sub>2</sub> samples. Table 1 indicates that the overpotentials at 10 mA/cm<sup>2</sup> and the Tafel slopes of prepared Fe<sub>0.75</sub>Co<sub>0.25</sub>S<sub>2</sub> and FeS<sub>2</sub> catalysts are equal to or even lower than that of the most of the reported transition metal/S-based OER electrodes, further implying the competitiveness of prepared electrocatalysts. Similarly, the FeS<sub>2</sub> had the smallest R<sub>ct</sub> value, further demonstrating the most accelerated charge transport kinetics during the OER process, Fig. 4d. Furthermore, the contact angle measurement was conducted to evaluate the penetration of the electrolyte into the FeS<sub>2</sub> electrode. The FeS<sub>2</sub> electrode exhibits a superhydrophilic behavior, with a contact angle approaching 0° (Fig. S5).

For practical application, it is crucial to estimate the stability of the electrocatalysts during the OER process. We carried out the chronopotentiometry at 1.547 V vs. RHE to identify the stability of the FeS<sub>2</sub> catalyst, Fig. 4e. Notably, the current density remained almost unchanged with an approach of 50 mA/cm<sup>2</sup> for 16 h. In addition, As shown in Fig. 4e, the electrocatalytic activity for FeS<sub>2</sub> after 1000 CV cycles changed little compared with that of initial LSV.

The C<sub>dl</sub> and R<sub>f</sub>, which can be determined by cyclic voltammetry (CV) measurements in the non-Faradic regions, are typically indexed to estimate electrochemical surface area (ECSA). According to the CV measurement for Ni foam, Fe<sub>x</sub>Co<sub>1-x</sub>S<sub>2</sub> in Fig. S3 at scan rates of 20, 40, 60, 80, 100 mv/s in 1 M KOH, the C<sub>dl</sub> for FeS<sub>2</sub> is 1.93 mF/cm<sup>2</sup>, which is much higher than those of 1.64, 1.62, 0.69, 0.38 and 0.24 mF/cm<sup>2</sup> for Fe<sub>0.75</sub>Co<sub>0.25</sub>S<sub>2</sub>, Fe<sub>0.5</sub>Co<sub>0.5</sub>S<sub>2</sub>, Fe<sub>0.25</sub>Co<sub>0.75</sub>S<sub>2</sub>, CoS<sub>2</sub> and Ni foam, respectively, indicating that FeS<sub>2</sub> possesses the highest ECSA among those various samples and number of active sites, further accelerating the OER process. Moreover, R<sub>f</sub> reflects the non-uniformity of the working electrode and a larger R<sub>f</sub> means more active surface area for driving the OER process. According to R<sub>f</sub> = C<sub>dl</sub>/60, the values of R<sub>f</sub> for FeS<sub>2</sub> is 32.1, higher than that of Fe<sub>0.75</sub>Co<sub>0.25</sub>S<sub>2</sub> (27.3), Fe<sub>0.5</sub>Co<sub>0.5</sub>S<sub>2</sub> (27), Fe<sub>0.25</sub>Co<sub>0.75</sub>S<sub>2</sub> (11.5), CoS<sub>2</sub> (6.3) and Ni foam (4) in terms of double-layer capacitance, respectively, implying the most active area for FeS<sub>2</sub>.

Considering the practical application of two-electrode electrolysis,



**Fig. 4.** Electrochemical performances of prepared samples for OER: (a) polarization curve (b) overpotentials at 10, 50, and 100 mA/cm<sup>2</sup> (c) Tafel slopes (d) Nyquist plots in the frequency range of 10<sup>5</sup>-0.1 Hz tested in 1 M KOH (e) chronopotentiometry of the FeS<sub>2</sub> catalyst performed under a constant potential of 1.547 V vs. RHE for 16 h and (f) polarization curve for FeS<sub>2</sub>/NF initial and after 1000 cycles after chronopotentiometry for 16 h.

**Table 1**

The OER activity of the prepared Fe<sub>x</sub>Co<sub>1-x</sub>S<sub>2</sub> (x = 0, 0.25, 0.5, 0.75 and 1) in this work, compared with reported transition metal/S-based OER electrodes in alkaline media (1 M KOH) reported in previous references.

Catalysts	$\eta$ /mV 10 mA cm <sup>-2</sup>	Tafel slope /mV dec <sup>-1</sup>	References
FeS <sub>2</sub>	189.5	71	Wang et al., 2021
CoFe-Co <sub>8</sub> FeS <sub>8</sub>	290	38	Wang et al., 2019
Co <sub>8</sub> FeS <sub>8</sub> /CoS	278	49	Wang et al., 2020a
Fe <sub>7</sub> S <sub>8</sub> /FeS <sub>2</sub> /C	262	48	Xu et al., 2021
FeCoS <sub>2</sub> /Co <sub>4</sub> S <sub>8</sub> /NFG	276	148	Wang et al., 2023c
Ni-Ni <sub>3</sub> S <sub>2</sub> @carbon NP	284.7	56	Lin et al., 2019
Co <sub>0.3</sub> Ni <sub>0.3</sub> Fe <sub>0.2</sub> S NPs/C	266	47	Wang et al., 2023a
Ni <sub>3</sub> S <sub>2</sub> /MoS <sub>2</sub>	260	59	Wang et al., 2020b
CoNi <sub>2</sub> S <sub>4</sub> @CoS <sub>2</sub>	259	45	Huang et al., 2019
Co <sub>9</sub> S <sub>8</sub> Hollow Spheres	285	58	Feng et al., 2017
CoS <sub>2</sub> HNNS	290	57	Ma et al., 2018
CoMoOS NBs	281	75.4	Xu et al., 2020
Fe <sub>1-x</sub> S/C	130	55	Wang et al., 2023b
CoS <sub>2</sub> -FeS <sub>2</sub>	210	46	Wang et al., 2020c
Fe <sub>0.75</sub> Co <sub>0.25</sub> S <sub>2</sub>	270	48	This work
FeS <sub>2</sub>	240	45	This work

the overall water splitting for FeS<sub>2</sub>/NF and Pt/C/NF (FeS<sub>2</sub>/NF//Pt/C/NF) served as an anode and a cathode was performed in 1 M KOH. Meanwhile, we assembled the RuO<sub>2</sub>/NF//Pt/C/NF and Ni foam//Ni foam two-electrode systems, displayed in Fig. 5b. Apparently, the FeS<sub>2</sub>/NF//Pt/C/NF system just required only 1.460 and 1.642 V to generate the current densities of 10 and 50 mA/cm<sup>2</sup>, respectively, which was much lower than those of RuO<sub>2</sub>/NF//Pt/C/NF (1.55 and 1.863 V) and Ni foam//Ni foam (1.86 V at 10 mA/cm<sup>2</sup>) systems, Fig. 5c. Interestingly, the FeS<sub>2</sub>/NF//Pt/C/NF system remained high stability at 123 mA/cm<sup>2</sup> for 16 h, Fig. 5d. All the above results indicates that FeS<sub>2</sub>/NF//Pt/C/NF two-electrode system possess high catalytic performance and stability toward the practical overall water splitting.

### 3.2. Characterization after OER

It is very crucial to get insight into the crystal phase, structure and morphology changes after OER for surveying the catalytic capability, stability and active site for OER catalysts. So, the SEM, TEM, HRTEM, XRD, Raman and XPS were performed to characterize the FeS<sub>2</sub> catalysts after OER. As shown in Fig. 6a and 6b, the FeS<sub>2</sub> samples after OER still kept the morphology of nanoparticles, further implying its stability. Interestingly, the nanoparticles after OER became much smaller, with a more uneven surface. The TEM and HRTEM in Fig. 6d-6g demonstrated the roughened surface caused by the OER process is very different from the smooth surface in Fig. 2. We also further inspect the diffraction of the catalysts after OER. After magnified in Fig. 6e and 6g, the single FeS<sub>2</sub> catalyst before OER had largely evolved into crystalline Fe<sub>2</sub>O<sub>3</sub> after OER, which could be verified by the (220) planes of FeS<sub>2</sub> and (110) planed of Fe<sub>2</sub>O<sub>3</sub>, matched with the lattice fringes of 0.19 and 0.25 nm, respectively, accompanying with some amorphous phase of FeOOH marked with a yellow circle. In addition, the mapping in Fig. 6c also showed the dispersed Fe, S and O elements.

In the Raman spectrum of FeS<sub>2</sub> after OER process in Fig. 7a, the peak at 555 cm<sup>-1</sup> corresponds to the vibration of FeOOH, implying the formation of amorphous CoOOH, while another broad peak around 1050 cm<sup>-1</sup> is the signal peak for \*OOH (Moysiadou et al., 2020; Koza et al., 2013). The main peaks of Fe, S and O had changed obviously, as shown in Fig. 7b. Comparing with Fe 2p before and after OER, it is obvious that the peaks at 708.0 and 720.8 eV belonging to Fe(II) had almost vanished in Fig. 7b (Guo et al., 2020). Meanwhile, the peaks at 711.2 and 724 eV in Fig. 7c belonging to Fe(III) also shifted to 711.6 and 724.4 eV, which could attribute to Fe-O and Fe-OOH bonds (Chen et al., 2022), further proofing the formation of Fe<sub>2</sub>O<sub>3</sub> and the amorphous phase of Fe-OOH. It happened that the peak intensities at 162.2 and 163.3 eV belonging to S 2p<sub>3/2</sub> and 2p<sub>1/2</sub> also became weaker and even disappeared for FeS<sub>2</sub> after OER, while the peak strength at 168.4 eV attributed to SO<sub>4</sub><sup>2-</sup> became stronger after OER (Wang et al., 2017), resulting from the oxidation for FeS<sub>2</sub>, as displayed in Fig. 7c. The O 2p spectrum after OER can be decomposed into Fe-O or Fe-OOH at 531.7 eV and adsorbed H<sub>2</sub>O at 533.5 eV, respectively (Abu-Zied and Ali, 2018; Yu et al., 2017), in

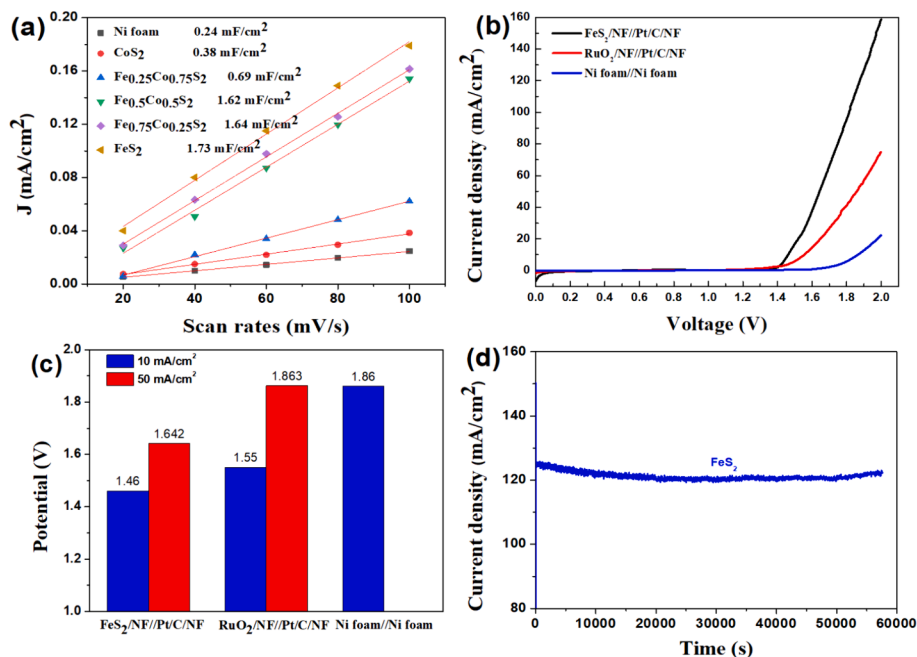


Fig. 5. (a) the plots of the current density v.s. scan rates of various samples, (b) the LSV curves, (c) overpotentials at 10 and 50 mA/cm<sup>2</sup> for two electrode water splitting of bare Ni foam//Ni foam, RuO<sub>2</sub>/NF//Pt/C/NF, and FeS<sub>2</sub>NF//Pt/C/NF systems in 1 M KOH. (d) i-t curves of FeS<sub>2</sub> for water splitting at 1.551 V.

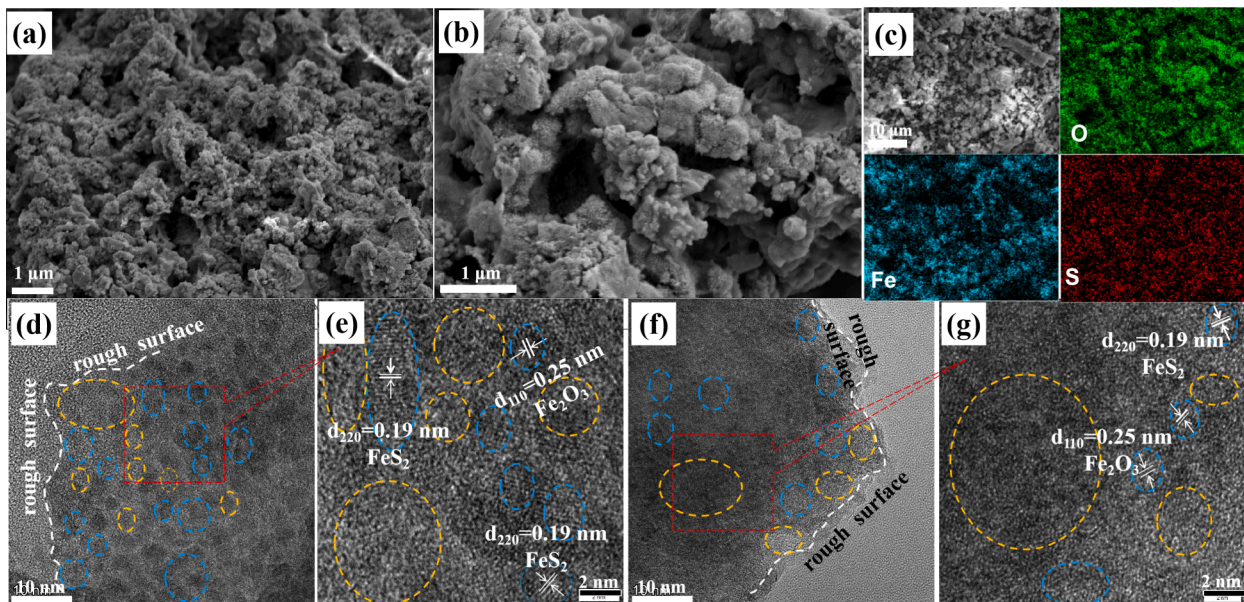


Fig. 6. (a-b) SEM, (c) elemental mapping (d, f) TEM, and (e, g) HRTEM of FeS<sub>2</sub> after OER.

Fig. 7d.

The EDX elemental analysis before and after OER was displayed in Fig. 7e. The relative content of S had reduced from 36.25 % to 1.77 % after OER. In contrast, the relative content of O increased from 3.28 % to 26.80 % after OER. In addition, the XRD patterns in Fig. 7f show that the obvious characteristic diffraction peaks for Fe<sub>2</sub>O<sub>3</sub> and FeS<sub>2</sub> appeared, indicating that the FeS<sub>2</sub> have largely evolved into crystalline Fe<sub>2</sub>O<sub>3</sub>, further suggesting the reconstruction of the catalyst. The peak at 44.67° can be attributed to the Ni, because the catalysts were dropped onto the foam Ni. All the above discussion indicates that the FeS<sub>2</sub> catalysts with smooth surface and uniform dispersion have largely reconstructed to crystalline-amorphous Fe<sub>2</sub>O<sub>3</sub>/Fe-OOH.

Based on the above discussion, the synergistic effect of Co contents

and reconstructing influence on Fe/S catalysts is put forward and is described in Scheme 2. First, the strong combination of Fe-S bond and the uniform dispersion of FeS<sub>2</sub> endow it improved charge transfer capability. During OER process, the FeS<sub>2</sub> with smooth surface has largely reconstructed to crystalline-amorphous Fe<sub>2</sub>O<sub>3</sub>/Fe-OOH with rough surface, which has a energy barrier for elementary reaction of O\*→OOH\*, further boosting the generation of O<sub>2</sub> (Su et al., 2024). As a result, the FeS<sub>2</sub> with fast electron transfer and low energy barrier for O\*→OOH\* exhibits excellent electrocatalytic water splitting capability towards OER.

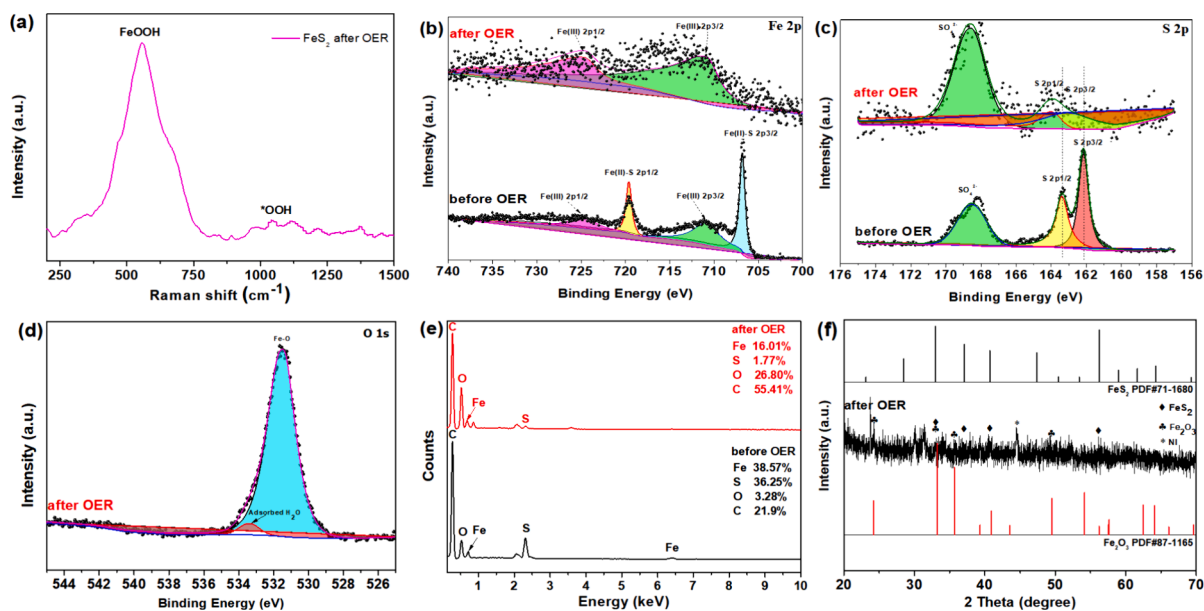
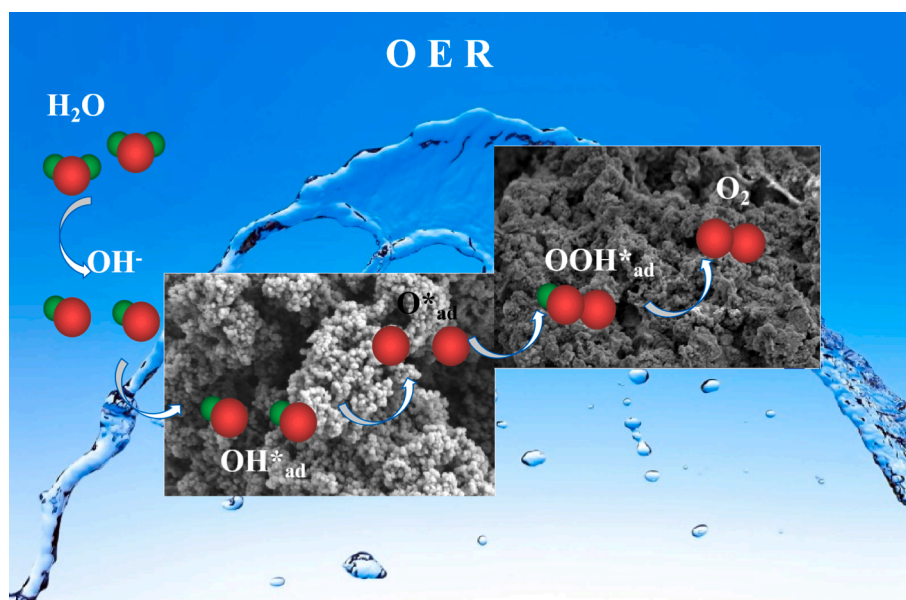


Fig. 7. (a) Raman spectrum of  $\text{FeS}_2$  after OER process, XPS spectra of (b) Fe 2p (c) S 2p (d) O 1s (e) the percentages of Fe, S, O, and C, (f) XRD patterns with  $\text{FeS}_2$  after OER.



Scheme 2. OER process in  $\text{FeS}_2$  electrocatalysts.

#### 4. Conclusion

In this study, highly-efficient  $\text{Fe}_x\text{Co}_{1-x}\text{S}_2$  ( $x = 0, 0.25, 0.5, 0.75$  and 1) nanoparticles toward OER in alkaline electrolyte were designed. Remarkably, the  $\text{FeS}_2$  delivered the most outstanding OER activity among prepared catalysts, with the overpotential of 240 mV for 10 mA/cm<sup>2</sup>, ultra-low Tafel slope of 45 mV/decades and high stability for 16 h. XRD, SEM, HRTEM, Raman and XPS characteristics before and after OER process, TOF and electrochemical results indicate that surface reconstruction, strong combination of F-S bonds and uniform dispersion of particles endow  $\text{FeS}_2$  with excellent catalytic property. The  $\text{FeS}_2$  with fast electron transfer and low energy barrier for  $\text{O}^* \rightarrow \text{OOH}^*$  exhibits excellent electrocatalytic water splitting capability towards OER. This study may provide a new idea for the research of metal sulfide electrocatalysts toward OER for alkaline water splitting systems.

#### CRediT authorship contribution statement

**Xiaozhen Ren:** Writing – review & editing, Writing – original draft, Investigation, Funding acquisition. **Shanshan Li:** Investigation, Formal analysis. **Ziyou Li:** Methodology, Investigation. **Zhenyang Zhang:** Methodology. **Hanxu Hou:** Methodology, Investigation. **Yanan Zhou:** Methodology, Investigation. **Chuanju Jin:** Methodology, Investigation.

#### Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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## Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.arabjc.2024.105907>.

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