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Arabian Journal of Chemistry

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ORIGINAL ARTICLE

# An insight to catalytic synergic effect of Pd-MoS<sub>2</sub> nanorods for highly efficient hydrogen evolution reaction



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Received 8 December 2021; accepted 24 January 2022

Available online 01 February 2022

## KEYWORDS

Transition metal dichalcogenides;  
Catalytic hydrogen production;  
Intrinsic activity;  
Electrical conductivity;  
Stability;  
Solvothermal

**Abstract** The electrocatalytic hydrogen evolution reaction (HER) is a sustainable energy production route using green chemistry. Transition metal dichalcogenides' application in catalytic hydrogen production is limited due to a lack of solutions that simultaneously address intrinsic activity, increased surface area, electrical conductivity, and stability problems. Herein we address these issues simultaneously by modifying the electronic structure of molybdenum disulfide (MoS<sub>2</sub>) nanorods using a low content of Pd (1 wt% and 2 wt%) dopant via a facile colloidal solvothermal route. The resulting MoS<sub>2</sub> nanorods doped with (1 and 2 wt%) palladium demonstrate current density of 100 mA/cm<sup>2</sup> at quite lower over-potentials of 137 mV and 119 mV than 273 mV for pure MoS<sub>2</sub>.

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nanorods, accompanied by high stability. This research proposes a strategy for designing high-performance HER electrocatalysts that work in acidic medium. In addition, the Tafel slope calculated for MoS<sub>2</sub> is 112 mV/dec whereas for 1 and 2 wt% Pd-MoS<sub>2</sub>, the Tafel slopes are 70 mV/dec and 46 mV/dec.

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## 1. Introduction

In the current economic scenario, we require more and more efficient technologies in every field due to increasing energy demand. The current high reliance on non-renewable fossil fuels (which are rapidly becoming unsustainable) has diverted the attention quickly and more deeply than in the past. As a result, scientists are putting out every effort to find a sustainable, renewable, and clean energy source in order to address energy concerns (Liling et al., 2020; Liling et al., 2019). The electrocatalytic hydrogen evolution process (HER:  $2\text{H}^+ + 2\text{e}^- \rightarrow \text{H}_2$ ), which is critical for producing clean hydrogen energy, is one such clean energy generating phenomena (Jingying et al., 2020; Zexing et al., 2012). The lack of a cost-effective alternative for Pt has hindered the scale-up of hydrogen electrochemical generation for decades; comparable catalytic compounds are largely confined either by low catalytic performance or a finite lifespan (Zexing et al., 2021; Seh et al., 2017; Yang, 2016). As a result, low-cost, high-stability, and high-natural-abundance catalysts are required, as well as catalysts that would not become obsolete and will not obstruct HER implementation. Until recently, Pt/C was indeed the gold standard catalyst for hydrogen production (Ma, 2014). Recent research suggests that Mo based materials such as MoO<sub>2</sub> could be a promising non-noble-metal electrocatalyst for HER (Baohua et al., 2017). MoS<sub>2</sub> has been considered as particularly favorable for hydrogen evolution reaction (HER) since the activity of its metallic edges (GH = 0.06 eV) was theoretically anticipated by Norskov and co-workers, and later experimentally confirmed by Jaramillo and co-workers (Hinnemann, 2005; Jaramillo, 2007). The following are the main governing concepts for optimizing MoS<sub>2</sub> catalytic efficiency: 1. increase the density of atomically under-coordinated absorption sites in the trigonal prismatic phase (2H) MoS<sub>2</sub> by revealing edge sites or introducing in-plane sulfur vacancies (SVs) (Karunadasa, 2012; Kibsgaard et al., 2012; Tsai, 2017). Though, unleashing the inherently high activity is restrained by the semi-conducting property of 2H-MoS<sub>2</sub>, where the charge transfer efficacy is restricted by an insufficiency of electrons at the chemical interface (Voiry, 2016; Li, 2015). Secondly, switching the MoS<sub>2</sub> phase from 2H to 1T, which is more conductive and thus more catalytically active (Kan, 2014; Lukowski, 2013). The active sites in 1T–MoS<sub>2</sub> are the S atoms in the basal plane; however, despite the substantially increased site density, these S sites have less acceptable hydrogen adsorption properties (GH = 0.17 eV) (Peng et al., 2021). Aside from these issues with 2H-MoS<sub>2</sub> and 1T–MoS<sub>2</sub>, one big challenge with both of these substances have poor stability, since defective 2H-MoS<sub>2</sub> has a high sulfur leaching rate and 1T–MoS<sub>2</sub> is intrinsically metastable (Longfei and Jan 2021; Geng, 2016). MoS<sub>2</sub> appears to be widely applicable to the HER only when the electrical conductivity, site density,

inherent activity, and stability concerns are all addressed at the same time. Owing to lower hydrogen adsorption energy of Molybdenum disulfide (MoS<sub>2</sub>) compared to noble metals (among which Pt is one), indicating that MoS<sub>2</sub>-based materials have a lot of potential as a replacement for Pt in the HER (Yan et al., 2014; Yipeng et al., 2019). The catalytic efficiency of MoS<sub>2</sub> could be improved even further by doping and/or substituting heteroatoms to lower the energy of hydrogen adsorption and modify the orbital structure of the valence and conduction bands, according to theoretical research (Zheng et al., 2015; Zexing et al., 2020). According to current study, doping MoS<sub>2</sub> with palladium (Pd) or transition metals like zinc (Zn) has been shown to boost HER activity (Song et al., 2021; Wu et al., 2019). Since it is commonly known, Pd and Mo have similar atomic radii. Therefore, we chose the element Pd as a dopant atom to examine its effects on MoS<sub>2</sub> based on the aforementioned concepts (see Table 1 and 2).

With the aforementioned background, this project aimed to synthesize Pd-MoS<sub>2</sub> nanorods (1&2 wt%) using one-pot solvothermal approach and the subsequent optimized electrochemical activity of Pd-MoS<sub>2</sub> for hydrogen evolution reaction.

## 2. Experimental section

The detail of materials used and fabrication of pure MoS<sub>2</sub> nanorods and Pd doped MoS<sub>2</sub> is described in detail below.

### 2.1. Materials

This experiment employed only analytical-grade chemical reagents. Molybdenum (VI) dioxide bis (acetylacetone) [MoO<sub>2</sub> (acac)<sub>2</sub>] and oleylamine (OAm, 70%) were purchased from Sigma-Aldrich. Dibenzyl disulfide [(PhCH<sub>2</sub>)<sub>2</sub>Se<sub>2</sub>, 95%] was purchased from Alfa Aesar. Absolute ethanol and toluene were obtained from Sinopharm Chemical Reagent Ltd. All chemicals were used in the experiments without further purification.

### 2.2. Preparation of MoS<sub>2</sub> nanorods

In a typical synthesis, 0.5 mmol of Molybdenum (VI) dioxide bis (acetylacetone) [MoO<sub>2</sub> (acac)<sub>2</sub>] and 0.5 mmol Dibenzyl

**Table 1** Average crystallite size (D) of MoS<sub>2</sub> nanostructures.

Composition	Average Crystallite Size "D" (nm)
Pure MoS <sub>2</sub>	78.719
Pd-MoS <sub>2</sub> -1 wt%	124.747
Pd-MoS <sub>2</sub> -2 wt%	453.116

Catalyst	Electrolytes	Overpotential (mv vs.RHE) @100 mA/cm <sup>2</sup>	reference
2 wt%Pd-MoS <sub>2</sub>	0.5 M H <sub>2</sub> SO <sub>4</sub>	119	This work
Se-MoS <sub>2</sub> /Co <sub>0.2</sub> Ni <sub>0.8</sub> Se <sub>2</sub>	Universal pH	122	2
Col-Fel-B	1 M KOH	355	4
MoP@PC	0.5 M H <sub>2</sub> SO <sub>4</sub>	> 200	7
MoS <sub>2</sub> NFs/rGO	0.5 M H <sub>2</sub> SO <sub>4</sub>	> 400	8
MoS <sub>2</sub> -40	0.5 M H <sub>2</sub> SO <sub>4</sub>	> 200	9
1 T- MoS <sub>2</sub>	0.5 M H <sub>2</sub> SO <sub>4</sub>	> 200	18
M- MoS <sub>2</sub>	0.5 M H <sub>2</sub> SO <sub>4</sub>	> 300	19
N-WS <sub>2</sub> /Ni <sub>3</sub> FeN	1 M KOH	140	37
MoS <sub>2</sub> /rGO/NiS-5	1 M KOH	212	36

disulfide [(PhCH<sub>2</sub>)<sub>2</sub>S<sub>2</sub>] were dispersed in 10 mL of oylamine (OAm) using ultra-sonication for 30 min. After mixing well, 0.05 mg of Borane *tert*-butylamine (BTB) was added to the mixture of MoO<sub>2</sub> (acac)<sub>2</sub> and (PhCH<sub>2</sub>)<sub>2</sub>S<sub>2</sub> then ultrasonicate further for 10 min. The above-mentioned mixture was autoclaved for 12 h at 280 °C in a 50 mL Teflon-lined autoclave. The as-obtained product was centrifuged at 7000 rpm and washed with toluene 6 times before being dried in a vacuum oven at 70 °C for 3 h.

### 2.3. Preparation of Pd-MoS<sub>2</sub> nanorods

The synthesis technique for Pd-MoS<sub>2</sub> is the same as for MoS<sub>2</sub>, with the exception that Pd (acac)<sub>2</sub> solution (1 wt% Pd in OAM) was used as per the specified MoS<sub>2</sub>: Pd mass ratio. Similarly, another sample was prepared by adding 2 wt% Pd solution into the mixture of MoO<sub>2</sub> (acac)<sub>2</sub> and (PhCH<sub>2</sub>)<sub>2</sub>S<sub>2</sub>.

## 3. Characterizations

### 3.1. Physical characterization

The as-obtained products were characterized using variety of techniques. A Philips X'pert PRO X-ray diffractometer with Cu K,  $\lambda = 1.54182 \text{ \AA}$ , a JSM-6700F, and a Hitachi H-7650 for SEM and TEM, correspondingly, were used to investigate the structure and surface morphology of the MoS<sub>2</sub> nanorods and Pd-MoS<sub>2</sub> (1 wt% and 2 wt%). The elemental composition and oxidation number of the composite elements were investigated using an X-ray photoelectron spectroscopy (XPS) VGESCA-LAB MKII X-ray photoelectron spectrometer with Mg as the excitation source. At ambient temperature, a JYLABRAM-HR Confocal Laser Micro-Raman spectrometer was used to examine the Raman spectra using a 514.5 nm argon laser.

### 3.2. Electrochemical measurements

Electrochemical measurements were performed in a three-electrode system on a CHI660E electrochemical workstation, using an Ag/AgCl (in saturated KCl solution) electrode as the reference electrode and a graphite rod as the counter electrode. The following is how the working electrodes were managed to

make: To begin, 5 mg of annealed catalyst (at 350 °C for 2 h in an inert environment) was dispersed in 1 mL of ethanol, and 40  $\mu\text{L}$  of catalyst ink was drop-cast onto carbon paper ( $0.5 \times 0.5 = 0.25 \text{ cm}^2$ ), then dried in a 70 °C oven. The linear sweep voltammetry (LSV) curves were obtained from 0 to  $-0.8 \text{ V}$  vs Ag/AgCl electrode in 0.5 M H<sub>2</sub>SO<sub>4</sub> solution at a scan rate of 5 mV/S and the cyclic voltammetry (CV) measurements were tested under the same condition except the scan rate was 50 mV/s. Electrochemical impedance spectroscopic (EIS) tests were taken at potential  $-0.35 \text{ V}$  at frequency ranging from 100 kHz to 0.1 Hz. For stability purpose, samples were continuously cycled for 1000 and 3000 cycles. Electrochemical active surface area (ECSA) was estimated by calculating Cdl ( $\Delta J$  vs different scan rates) in non-faradic potential range 0.33–0.43 V.

## 4. Result and discussion

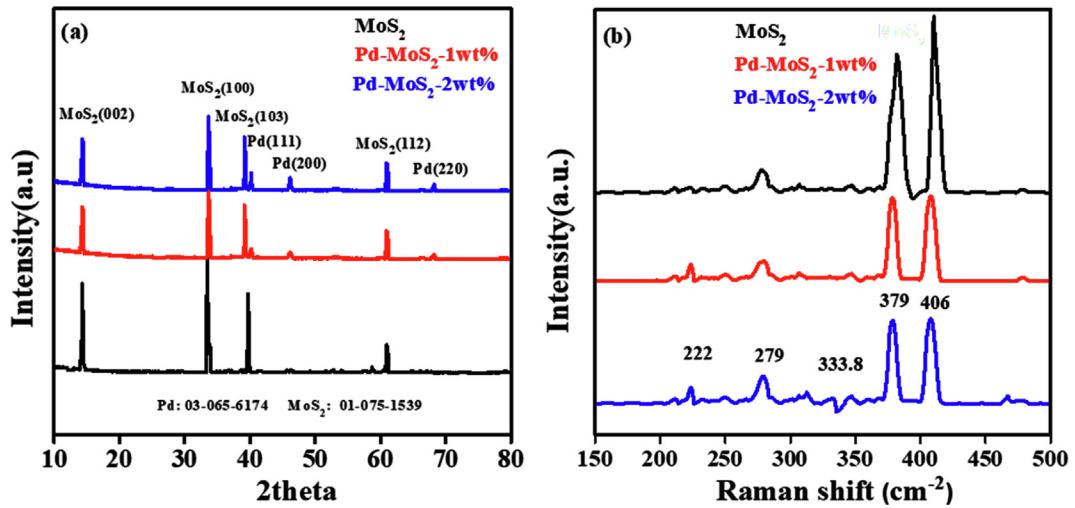
The crystallinity and phase study of pure MoS<sub>2</sub> and Pd doped MoS<sub>2</sub> nanorods (1 wt% & 2 wt%) were demonstrated using XRD technique. Each 1 wt% Pd-MoS<sub>2</sub> and 2 wt% Pd-MoS<sub>2</sub> have a set of distinctive peaks at 40.7°, 46.6°, and 68.1°, which correspond to the (111), (200), and (220) planes of the conventional fcc (face-centered-cubic) Pd crystalline structure, respectively (card # 03-065-6174) (Santra et al., 2013). In addition to the peaks for Pd, other peaks in the XRD pattern of Pd-MoS<sub>2</sub> that are simultaneously indexed to MoS<sub>2</sub> (002), (100), (103), and (112) demonstrate the effective doping of Pd on MoS<sub>2</sub> (card # 01-075-1539). Moreover, the XRD pattern of pure MoS<sub>2</sub> only demonstrate the peaks for its hexagonal phase planes (Zhong et al., 2019; Iqbal et al., 2020). Scherer's formula is used to compute the crystalline size of the manufactured MoS<sub>2</sub> nanostructures.

$$D = \frac{K\lambda}{\beta \times \cos\theta} \quad (1)$$

where  $\beta$  corresponds to FWHM value and  $\theta$  represents angle of diffraction.

Raman spectroscopy is an effective tool that can detect material thickness, crystallographic orientation and phase transition. The comparative Raman spectra of as-grown MoS<sub>2</sub> nanorods and Pd-MoS<sub>2</sub> ((1 wt% & 2 wt%) are shown in Fig. 1b. The pure MoS<sub>2</sub> nanorods display two distinct Raman peaks, the E<sub>2g</sub><sup>1</sup> mode at 379 cm<sup>-1</sup> and the A<sub>1g</sub> mode at 410 cm<sup>-1</sup>, as shown in the figure. These findings are in agreement with previous studies (Late et al., 2014). The deposition of Pd atoms in nanorods causes an apparent red shift of around 4.0 cm<sup>-1</sup> of the A<sub>1g</sub> peak, whereas the position of the E<sub>2g</sub><sup>1</sup> peak nearly remains unchanged. As a consequence, the difference ( $\Delta$ ) between both the E<sub>2g</sub><sup>1</sup> and A<sub>1g</sub> peaks in pure MoS<sub>2</sub> nanorods drops from 31.0 to 27.0 cm<sup>-1</sup> in Pd-MoS<sub>2</sub> nanorods (1 wt% & 2 wt%) (Lin et al., 2014). As a result, this demonstrates that Pd doping can produce a considerable modification in the electronic structure of MoS<sub>2</sub> nanorods. Furthermore, the two new weak peaks found in Raman spectra of Pd-MoS<sub>2</sub> (1 wt% & 2 wt%) at 222 cm<sup>-1</sup> and 33.8 cm<sup>-1</sup>, which are coincided with the 1T phase molybdenum disulfide, could be attributed to modifications in the material's structure. The weak peak at 283 cm<sup>-1</sup> belong to E<sub>1g</sub> ([33]).

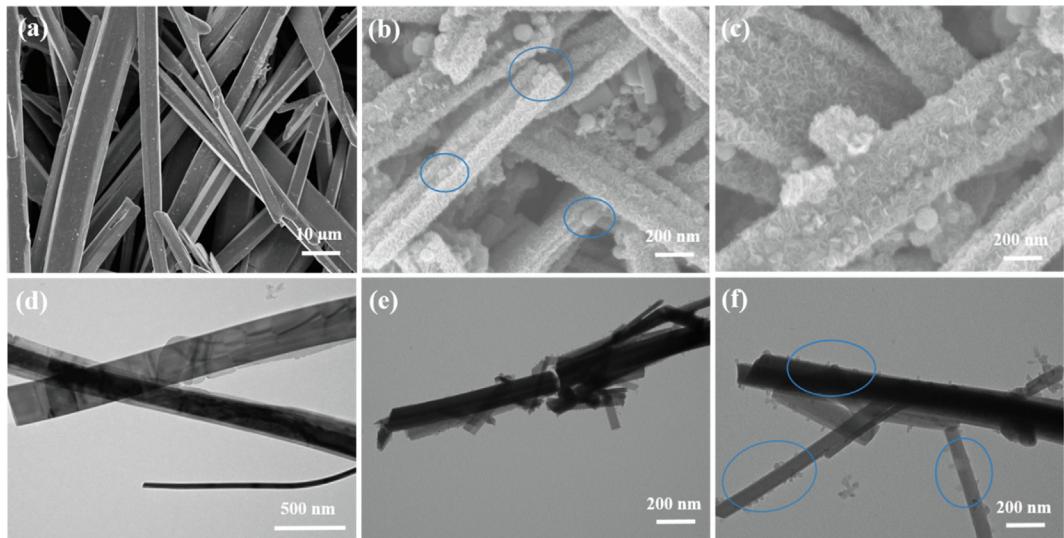
In Fig. 2, SEM and TEM micrographs were used to examine the microstructure and morphology of MoS<sub>2</sub> nanorods and



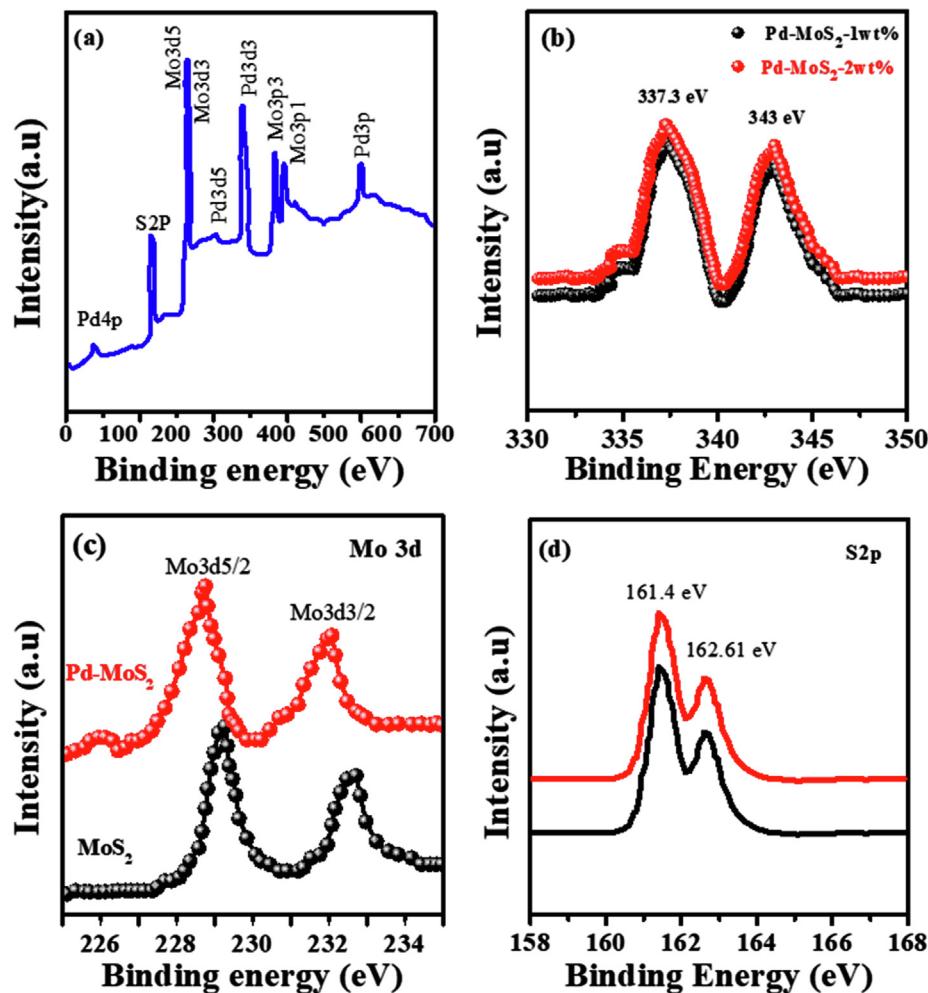
**Fig. 1** (a) XRD pattern of MoS<sub>2</sub>, Pd-MoS<sub>2</sub> (1 wt% & 2 wt%), (b). Raman spectra of MoS<sub>2</sub>, Pd-MoS<sub>2</sub> (1 wt% & 2 wt%).

Pd-MoS<sub>2</sub>. SEM and TEM images of pure MoS<sub>2</sub> nanorods revealed rod-like morphology, as illustrated in Fig. 2a and 2d. The numerous tiny Pd NPs distributed over the substrate on MoS<sub>2</sub> nanorods for Pd-MoS<sub>2</sub> (1 wt%) are clearly visible in SEM and TEM micrographs (Fig. 2b and e). More intriguingly, as demonstrated in transmission electron microscopy (TEM) image Fig. 2f, apparent Pd particles were regularly produced on the MoS<sub>2</sub> nanorods. Micrographs of the Pd-MoS<sub>2</sub> (2 wt%) nanorods using scanning electron microscopy (Fig. 2c) and transmission electron microscopy (Fig. 2f) reveal that Pd particles are uniformly adorned on the surfaces, as seen in the images encircled. The characteristic XPS spectrum of Pd-MoS<sub>2</sub> nanorods is shown in Fig. 3a, demonstrating the coexistence of Pd, Mo, and S. Pd element binding energies in the Pd doped MoS<sub>2</sub> nanorods are shown in Fig. 3b. Two distinctive peaks at 337.3 eV and 343.0 eV are attributed to Pd

3d<sub>5/2</sub> and 3d<sub>3/2</sub>, correspondingly. The binding energies are substantially higher than those of Pd metals, and they are strikingly similar to those of Pd<sup>4+</sup> (Yuwen et al., 2014; Hao et al., 2015). This implies that Mo atoms have been successfully replaced by Pd dopants, and that the Pd atoms have been covalently stabilized inside the lattice. Fig. 3c and d typically demonstrate high resolution XPS spectra of Mo 3d and S 2p. The binding energies of Mo and S for the MoS<sub>2</sub> nanorods without Pd doping are depicted in black curves. The Mo 3d<sub>5/2</sub> and Mo 3d<sub>3/2</sub> orbitals are ascribed to the peaks at 229.2 and 232.6 eV, correspondingly. When compared to the pure MoS<sub>2</sub> nanorods, the fundamental peaks of Mo in the Pd-MoS<sub>2</sub> nanorods reveal a smooth change toward lower binding energies, as seen in the figure. This shift could be explained by a decrease in Fermi level (EF) as a result of p type doping, as previously seen in other research (McDonnell et al., 2014). The



**Fig. 2** (a) SEM image of MoS<sub>2</sub> nanorods (b & c) SEM images of Pd-MoS<sub>2</sub> (1 wt% & 2 wt%), (d) TEM image of MoS<sub>2</sub> (e, f) TEM image of Pd-MoS<sub>2</sub> (1 wt% & 2 wt%).



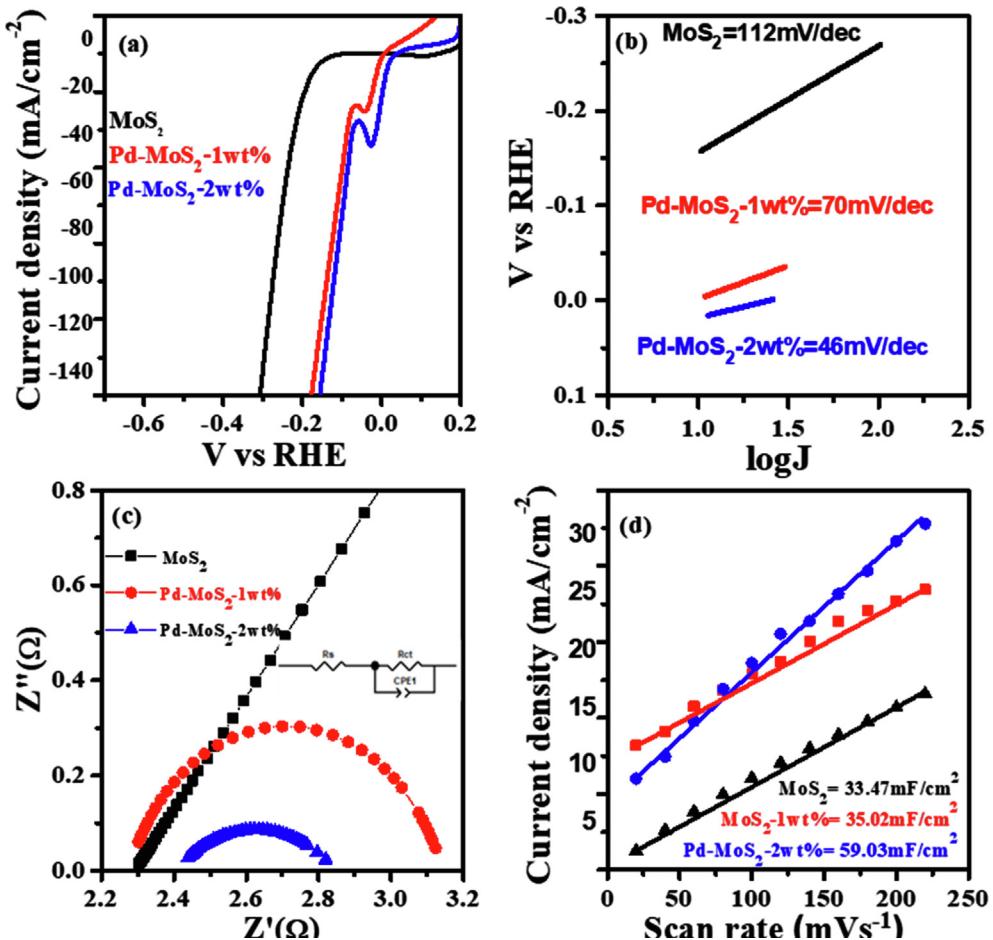
**Fig. 3** (a) XPS survey scan of Pd-MoS<sub>2</sub> nanorods, (b) Pd 3d (c) Mo 3d (black for MoS<sub>2</sub> nanotube and red for Pd-MoS<sub>2</sub> 2 wt%), (d) S2p (black for MoS<sub>2</sub> nanotube and red for Pd-MoS<sub>2</sub> 2 wt%).

binding energies of the S 2p<sub>1/2</sub> and S 2p<sub>3/2</sub> emerge at 163.5 and 162.2 eV, correspondingly, and there is essentially no change between the nanorods with and without Pd doping.

The HER catalytic behavior of MoS<sub>2</sub> and Pd-MoS<sub>2</sub> was investigated using the demonstrative linear sweep voltammograms (LSVs) shown in Fig. 4a. The electrode with pure MoS<sub>2</sub> exhibited the lowest catalytic activity in acidic medium, which can be seen in Fig. 4a. Both the 1 wt% and 2 wt% Pd-MoS<sub>2</sub> samples displayed increased catalytic activity and lower overpotential when compared to pristine MoS<sub>2</sub>. Initially, the pristine MoS<sub>2</sub> had a current density of 100 mA/cm<sup>2</sup> at 273 mV, which was analogous to prior MoS<sub>2</sub>/rGO/NiS observations (Guangsheng et al., 2021). Secondly, 1 Wt% Pd doping on MoS<sub>2</sub> ( $\eta$ @100 mA cm<sup>-2</sup> = 137 mV) further improve the activity. Further, Pd doping led to a dramatic improvement in catalytic performance toward the HER, which considerably outperforms 1% Pd-MoS<sub>2</sub> and pure MoS<sub>2</sub> catalysts. The 2 Wt% Pd-MoS<sub>2</sub> catalyst displayed a current density of 100 mA/cm<sup>2</sup> at an overpotential of only 119 mV, which is significantly better than that of hetero-structured tungsten-based catalysts (Jinsong et al., 2022). This is the best performance ever for heteroatom-doped MoS<sub>2</sub>-based catalysts in acidic media (Zhilong et al., 2020; Luozhen et al., 2018; Sun et al.,

2018). Tafel slope is key descriptor to evaluate the kinetics of HER. The Tafel slope was used to assess the kinetic behaviour of the Pd-MoS<sub>2</sub> nanorods for HER. As shown in Fig. 4b, the Tafel slope of 2 Wt% Pd-MoS<sub>2</sub> nanorods is 48 mV/dec, which is lower than 1 Wt% Pd-MoS<sub>2</sub> (70MmV/dec), MoS<sub>2</sub> nanorods (112 mV/dec) and the majority of previously described MoS<sub>2</sub>-based HER electrocatalysts implying that Pd-MoS<sub>2</sub> nanorods has better reaction kinetics in the HER process (Zheng et al., 2018; Meng, 2020). The Tafel slope indicated a gradual drop from 112 mV/dec to 70–48 mV/dec, demonstrating that these materials have HER kinetics based upon Heyrovsky reaction, which is the rate-determining step in electrochemical desorption (Jung et al., 2017; Jue et al., 2019).

Moreover, Electrochemical impedance spectroscopy (EIS) measurements at -350 mV vs reversible hydrogen electrode (RHE) from 100 kHz to 0.1 Hz were performed to better investigate electrode kinetics during the HER process. Fig. 4c illustrates the Nyquist plots of these electrodes and the fitted equivalent circuit in the inset. The smaller and larger semicircles indicate the electrode-electrolyte interface resistance and the charge transfer resistance (Rct) of the H<sup>+</sup> reaction, respectively (Luo et al., 2018). As demonstrated in Fig. 4c, when compared to pure MoS<sub>2</sub> nanorods, the charge-transfer



**Fig. 4** (a) Polarization curves of pure MoS<sub>2</sub> nanorods, Pd-MoS<sub>2</sub> (1 wt% & 2 wt%), (b) Tafel slopes (c) Nyquist plot (d) Double layer capacitance recorded in non-faradic area of successive LSVs.

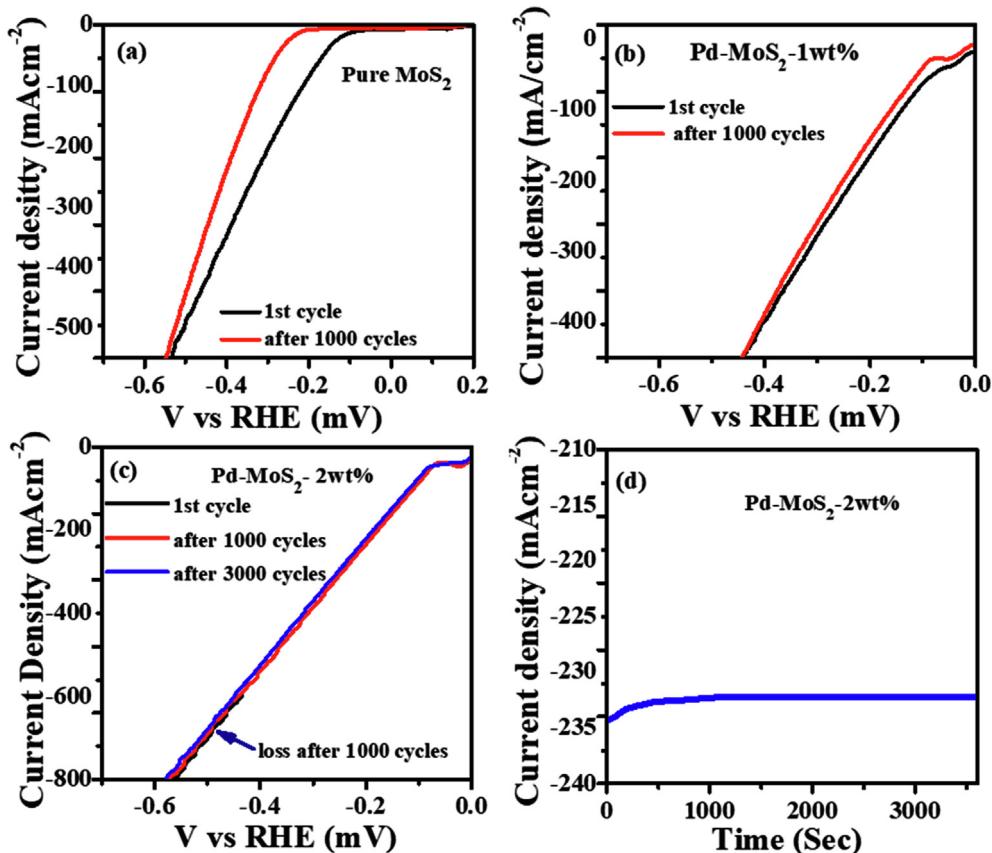
resistances drop significantly, with 1 Wt% Pd-MoS<sub>2</sub> (1.05) and 2 Wt% Pd-MoS<sub>2</sub> (0.4) having the lowest  $R_{\text{ct}}$  values. These findings suggest that a significant concentration of Pd doping can greatly enhance conductivity and charge transfer, hence boosting reaction efficiency. To gain a better understanding of these electrocatalysts, the electrochemical surface area (ECSA) values were examined. The electrochemical double layer capacitances ( $C_{\text{dl}}$ ) were calculated using conventional CV, and the typical CV curves recorded for the different catalyst with variable scan rates are presented in Fig d (Fengming et al., 2021; Li et al., 2020; Miguel et al., 2015). In this scenario, a greater  $C_{\text{dl}}$  value indicates a higher ECSA. The  $C_{\text{dl}}$  of the 2 Wt% Pd-MoS<sub>2</sub> catalyst was approximately  $59.03 \text{ mF cm}^{-2}$ , substantially higher than that of the 1 Wt% Pd-MoS<sub>2</sub> ( $35.02 \text{ mF cm}^{-2}$ ) and pure MoS<sub>2</sub> ( $33.47 \text{ mF cm}^{-2}$ ) as indicated in Fig. 4d. This result underpinned the effective synthesis of the Pd-MoS<sub>2</sub> catalyst, in which the 2Wt% doping of Pd on MoS<sub>2</sub> nanorods exposed more active sites 1 Wt% Pd-MoS<sub>2</sub> and pure MoS<sub>2</sub> nanorods. The Pd doped MoS<sub>2</sub> catalyst's HER performance was improved as a result of its efficient surface charge transfer, faster reaction rate, and large ECSA.

We are concerned about the catalyst's electrochemical stability because the hydrogen generating process could result in structural deformation and activity loss with continuous cycling (Ying et al., 2021; Zhijuan et al., November 2021). The electrocatalytic stability of Pd-MoS<sub>2</sub> nanorods was inves-

tigated by comparing LSV curves before and after cyclic voltammetry test (1000 cycle and even 3000 cycles for 2 wt% Pd-MoS<sub>2</sub>) in 0.5 M H<sub>2</sub>SO<sub>4</sub> solution. After 3000 cycles, the polarization curve shows no apparent change, as illustrated in Fig. 5c. Under the same experimental conditions, however, the HER stability of the pure MoS<sub>2</sub> nanorods as well as 1 wt% Pd-MoS<sub>2</sub> electrocatalysts (after 1000Cv cycles) is relatively poor. In electrocatalysis, durability is essential since the commercial electrode would have to be able to maintain a constant current for an extended period of time. To assess the durability, galvanostatic measurements were taken at potential of  $-0.4 \text{ V}$  and the result (Fig. 5d) demonstrated that Pd-MoS<sub>2</sub> operated well after 10 h of continuous operation.

## 5. Conclusions

In conclusion, using a facile solvo-thermal route, we were able to successfully fabricate Pd-MoS<sub>2</sub> electrocatalyst with precise composition (doping) and morphologies (nanoparticles on nanorods). Ultimately, we discovered a catalyst for the HER in acidic conditions that has superior catalytic activity, increased conductivity, high surface area, and long-term durability. This research could pave the way for more logical design of cost-effective and high-performance hydrogen evolution catalysts.



**Fig. 5** (a) Stability tests for the first LSV curve and one after 1000 CV cycles for MoS<sub>2</sub> nanorodes (b, c) Pd-MoS<sub>2</sub> (1 wt% and 2 wt% (upto 3000 Cv cycles)) (d) Time dependent i-t curves at -0.4 V.

## Acknowledgement

1. This work was financially supported by the National Natural Science of China (Nos. and) and Anhui Provincial Natural Science (No.).
2. Taif University Researchers Supporting Project number (TURSP-2020/241) Taif University, Taif Saudi Arabia.

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