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Highly efficient UV-visible absorption of TiO_2/Y_2O_3 nanocomposite prepared by nanosecond pulsed laser ablation technique



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KEYWORDS

Pulsed laser ablation in liquid; Nanocomposites; Titanium oxide; Yttrium oxide; Optical properties Abstract Nanostructured TiO₂-based composites are promising materials because of their superior optical, structural, and electronic properties relative to pristine nanostructured TiO₂. The enhanced properties of TiO₂-based composites have been used in several important applications such as gas sensors, solar cells, and photocatalytic applications. In the past, numerous materials have been coupled with TiO₂ to enhance their optical properties. In this work, full-spectrum (UV and Visible) responsive TiO₂ /Y₂O₃ nanocomposite has been synthesized via pulsed laser ablation in liquid (PLA) to study the impact of Y₂O₃ on the structural, morphology, and optical property of the TiO₂. The nanostructured composites prepared were characterized by XRD, Raman spectroscopy,

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Field-Emission Scanning Electron Microscope (FESEM) attached with Energy-Dispersive X-ray spectroscopy (EDX), Photoluminescence, XPS, and UV–Vis absorbance spectra. The result demonstrates that the coupling Y_2O_3 with TiO₂ not only changes the structural, optical, and morphology of the TiO₂ but also significantly amplified the light absorption characteristics of the TiO₂ within the UV and visible region. The synthesized TiO₂ /Y₂O₃ nanocomposite could potentially be useful for visible-light responsive applications.

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1. Introduction

Titanium dioxide (TiO₂) is a well-known semiconductor material that has generated significant research interest due to its wide range of applications in photo remediation (Daghrir et al., 2013), water splitting (Nguyen et al., 2020), optical (Inpor et al., 2008), electronic (Phani and Santucci, 2006), sensors (Mele et al., 2021; Tian et al., 2021), and solar applications (Almomani et al., 2022). The use of TiO₂ for various applications is due to its unique electronic and optical properties (Daghrir et al., 2013; Mele et al., 2021; Tian et al., 2021). TiO₂ is non-toxic, hence it is useful in cosmetics(Dréno et al., 2019), biomedical (McNamara and Tofail, 2016), and anti-cancer applications (Elsayed et al., 2022b). TiO₂ exists in four well-known polymorphs, namely, anatase (tetragonal), rutile (tetragonal), brookite (orthorhombic), TiO₂ (B) (monoclinic) (Siddiqui, 2020). The nanostructured anatase TiO_2 has received considerable interest in photocatalysis (Verbruggen, 2015) and dye-sensitized solar cells (Akila et al., 2019) research. Meanwhile, mixed polymorphs of TiO₂ -20% rutile and 80% anatase are more efficient for biomedical applications (Jafari et al., 2020). Different TiO₂ morphologies are appropriate for specific applications (Fahad et al., 2018; Gondal et al., 2015; Nabi et al., 2020). As such, several TiO₂ morphologies have been prepared for different applications. Some of the common TiO₂ morphologies include the quantum dot (Cui et al., 2014), nanoparticles, nanorods (Atabaev et al., 2016), nanoflowers (Dong et al., 2017), nanotubes(Cui et al., 2014), nanofibers (Kumar et al., 2007), etc., These morphologies are to a large extent a function of synthesis techniques used in the fabrication of nanostructured TiO₂. The literature reflects that the following techniques have been used in the preparation of TiO2-based materials; sol-gel, hydrothermal, solvothermal, direct oxidation, chemical vapor deposition, physical vapor deposition, electrodeposition, etc (Chen and Mao, 2007).

The suitability of TiO_2 for diverse application correlates with its wide and tunable electronic bandgap which range from 3.2 to 3.35 eV. Thus, permitting appropriate bandgap engineering for different applications (Chen and Mao, 2007). Studies have shown that the electronic bandgap is influenced by particle size, the presence of impurities, the shape of materials, surface charges, and phase transition (Mele et al., 2021; Tian et al., 2021), (Chen and Mao, 2007).

Despite the promising versatility of TiO₂, pure TiO₂ shows less promising results due to its high energy bandgap (Eg \sim 3.6 eV), which limits its functionality to mainly absorb ultraviolet (UV) light which represents only 4% of the solar spectrum. Therefore, to maximize the absorption of solar light and transportation of charge, TiO₂ has been doped and coupled with several nanomaterials such as Carbon/TiO₂ (Irie et al., 2003), Pt/TiO₂ (Yu et al., 2010), CdSe - TiO₂ (Kongkanand et al., 2008), TiO₂–Graphene (Wang et al., 2009), Graphene Oxide/TiO₂ (Chen et al., 2010), TiO₂/Au (Chen et al., 2010) for various applications. Despite the progress made in developing novel TiO₂-based nanocomposites, there are still materials with tremendous capacity that have not been properly explored. One such material is yttrium oxide, which is an air-stable and solid material that is used for various applications such as the synthesis of inorganic compounds, microwave filters, and ultrafast sensors used in gamma-ray and X-rays.

It is important to state that studies on Y_2O_3/TiO_2 nanocomposite are limited in the literature. A few examples of these studies are highlighted as follows: (Jun-ping and Ping, 2002) prepared Y_2O_3/TiO_2 catalysts by the impregnation method, and the catalysts were used for the decomposition of sodium dodecylbenzene sulphonate. The report concluded that the photocatalytic activity of the Y_2O_3/TiO_2 catalyst is about 2.4 times better than TiO₂. Also, (Ravishankar et al., 2016) prepared Y₂O₃/TiO₂ nanocomposite photocatalysts via a conventional hydrothermal method $(Y_2O_3/TiO_2 NC_{(HM)})$ and an ionic liquid assisted hydrothermal method (Y₂O₃/TiO₂). Both materials were assessed for photocatalytic hydrogen production via water splitting. At 25% optimized weight, the $Y_2O_3/$ TiO₂ NC exhibits a 2-fold- improvement in the photocatalytic performance relative to 25 wt% Y₂O₃/TiO₂. The study highlights that the synthesis approach is very important in finetuning the property of Y_2O_3/TiO_2 nanocomposite which may affect the application performance.

In general, the literature survey revealed that both fundamental and applied research on Y_2O_3/TiO_2 nanocomposite is lacking, as such, this present study is motivated by the gaps in research on the Y_2O_3/TiO_2 nanocomposite. As noted by (Ravishankar et al., 2016), the synthesis approach affects the performance functionality of the prepared Y_2O_3/TiO_2 nanocomposite. Hence, it is highly desirable to investigate the property of the Y_2O_3/TiO_2 nanocomposite prepared by robust technique to uncover new information that could potentially be useful for different applications. In this contribution, we report the synthesis of TiO_2/Y_2O_3 nanocomposite via pulsed laser ablation technique (PLA) for the first time. Further, we carried out thorough characterizations of the nanocomposite to gain better insight into the science of the material synthesized.

Interestingly, the result obtained shows a significant improvement in the absorbance characteristics which span the UV-visible region, which clearly shows a major improvement in the prepared nanocomposite relative to the previous studies where the absorbance enhancement is limited to the visible region. The PLA offers a green method for the synthesis of



Fig. 1 Experimental steps used to fabricate TiO₂/P25-Y₂O₃ nanocomposites via PLA.

nanostructured materials. In PLA, a high-power nanosecond pulsed laser is used to ablate the composites in liquid and as a result, a nanostructured composite of TiO_2/Y_2O_3 is formed. The nanomaterial formed via PLA is highly pure which might not be easy to produce in many other chemical synthesis routes. More importantly, the approach does not depend on the use of a surfactant or any other chemical additives, thus, this safety component is very useful for the preparation of nanomaterials for biomedical applications (Drmosh et al., 2010; Elsayed et al., 2022c, 2022a; Gondal et al., 2012, 2010).

2. Experimental work

The laser ablation in liquid setup has been used to prepare the TiO_2/Y_2O_3 composite as shown in Fig. 1. A high purity TiO_2 and Y₂O₃ (99.99%) were used as target materials. The two powders were purchased from Aldrich. 100 mg from each powder was dispersed separately in a beaker filled with 25 mL of deionized water and sonicated for 1 h. Firstly, the TiO2 colloidal solution was irradiated by a focused beam of a nanosecond pulsed Nd: YAG laser that operates at 266 nm, 30 m J, 10 ns, and 10 Hz for one hour. The laser beam was focused using a UV lens with a 200 mm focal length. The focus of the laser beam was adjusted under the surface of the liquid to avoid any high fluence that might cause ablation on the surface-air interface and to avoid the splashing of the liquid. The spot size on the liquid surface was kept at almost 5 mm. The colloidal solution was stirred at room temperature with a magnetic stirrer during the laser irradiation. Secondly, the colloidal solution of TiO2 was replaced by Y2O3 colloidal solution and irradiated under the same conditions. Finally, the two irradiated colloidal solutions of both TiO₂ and Y₂O₃ were mixed, and then the mixture was irradiated with the focused laser beam for one hour to fabricate TiO_2/Y_2O_3 nanocomposite.

3. Characterization techniques

The as-fabricated nanostructured materials were characterized via various techniques to reveal the morphology, composition,

and microstructure. Field Emission Scanning Electron Microscope (FE-SEM, Lyra3, Tescan) with an accelerating voltage of up to 20 kV coupled with energy-dispersive X-ray spectroscopy (EDX) silicon drift detector (X-MaxN, Oxford Instruments) was used to identify the morphology of the fabricated samples. X-ray powder diffraction (XRD, Rigaku MiniFlex) was utilized with Cu K α 1 radiation ($\lambda = 0.15416$ n m), an accelerating voltage of 30 kV, and a tube current of 10 mA to accurately identify the crystalline properties of the samples. The Raman spectroscopy was carried out by a Lab-Ram HR evolution Raman Spectrometer Horiba Scientific at room temperature with a 633 nm laser light. The photoluminescence (PL) measurements were conducted at room temperature at an excitation wavelength of 350 nm by a Fluorolog-3 spectrofluorometer system (Horiba Jobin-Yvon) in the emis-



Fig. 2 XRD diffraction patterns (a) $TiO_2/P25$, (b) Y_2O_3 , and (c) $TiO_2/P25$ - Y_2O_3 nanocomposites fabricated by PLA in water and their corresponding standards. XRD peaks of rutile and anatase phases are denoted by R and A, respectively.



Fig. 3 (a) Raman spectra, and (b) zoomed-in view of the TiO₂/P25, Y₂O₃, and TiO₂/P25-Y₂O₃ nanocomposites fabricated by PLA.



Fig. 4 FESEM images of (a) $TiO_2/P25$, (b) Y_2O_3 , and (c and d) low and high magnification of $TiO_2/P25-Y_2O_3$ nanocomposites fabricated by PLA.

sion spectral range 350–600 nm. The chemical composition of the nanocomposite was assessed using an ESCALAB 250Xi X-ray photoelectron spectroscope (XPS) that has a binding energy resolution of \pm 0.1 eV. The UV–Vis spectrophotometer

(Model SolidSpace-3700) was used to measure the optical properties of the synthesized materials using 10 mm quartz. The range of 200 - 900 nm was used to record the absorbance spectra.



Fig. 5 EDX mapping analysis of TiO₂/P25-Y₂O₃ nanocomposites synthesized by PLA.

4. Results and discussions

The XRD patterns of the TiO₂/P25, Y₂O₃, and TiO₂/P25-Y₂O₃ nanocomposites fabricated by PLA in water are shown in Fig. 2 a, b, and c, respectively. As shown in Fig. 2a, $TiO_2/$ P25 patterns reveal the co-existence of rutile (ICDD-PDF #21-1276) and tetragonal anatase (ICDD-PDF # 21-1272) phases. The diffraction peaks of anatase TiO₂ were found at $2\theta = 25.78^{\circ}, 36.58^{\circ}, 37.53^{\circ}, 38.32^{\circ}, 48.55^{\circ}, 53.89^{\circ}, 63.22^{\circ},$ 68.76°, and 75.58° which are indexed to (101), (103), (004), (112), (200), (105), (211), (204), and (116) crystal planes of anatase phase of TiO₂ (JCPDS #21-1272). Meanwhile, for the rutile TiO₂ the 2 θ are observed at 27.94°, 36.58°, 41.23°, 44.05°, and 56.64° (ICDD-PDF #21-1276) (Guo et al., 2017; Yao et al., 2020, 2017). The prominent peaks of the rutile TiO₂ were identified at 27.94 and 36.58 which correspond to the (110) and (101), crystal planes, respectively. The diffraction peaks observed in Fig. 2b at $2\theta = 20.94^{\circ}$, 29.61°. 34.28°, 36.37°, 40.38°, 43.98°, 47.43°, 49.03°, 50.59°, 53.73°, 56.67°, 58.09°, 59.53°, 60.91°, 64.99°, 71.50°, and 79.09° corresponded to cubic Y_2O_3 (ICDD-PDF #01-074-1828) (Benammar et al., 2020; Wang et al., 2013). The XRD pattern of $TiO_2/P25-Y_2O_3$ (Fig. 2c) matches well with both Y_2O_3 and TiO₂/P25, suggesting the successful integration of the two compounds. Furthermore, no other diffraction peaks except for that of the Y2O3 and TiO2/P25 phase were observed, indicating the high purity of the TiO₂/P25-Y₂O₃ nanocomposites sample and the successful integration of the two compounds. There is a reduction in the (101) peak belonging to the anatase TiO₂ in the TiO₂/P25-Y₂O₃ nanocomposites. The size of the nanoparticles is calculated via Debye Scherrer's formula $D = k \lambda/\beta \cos\theta$ (Cullity, 1956), where k, λ , β and θ are a constant (0s94), the X-ray wavelength (0s154016 nm), the full wavelength at half maximum, and the half diffraction angle, respectively. The result demonstrated that the crystal size of the TiO₂/P25 nanoparticles fabricated via PLA and the TiO₂/P25 nanoparticles in the TiO₂/P25-Y₂O₃ nanocomposites sample is about 21 ± 5 nm. The average crystal size of the Y₂O₃ synthesized using PLA and the Y₂O₃ in the TiO₂/P25-Y₂O₃ nanocomposites sample is about 39 ± 10 nm.

To further investigate the structural properties of the fabricated samples, Raman spectra were performed, and the obtained results are displayed in Fig. 3a and b. The nanostructured TiO₂/P25 sample showed Raman bands at 132 cm⁻¹ (E_g), 184 cm⁻¹ (E_g), and 382 cm⁻¹ (B_{1g}), which could be ascribed to the O–Ti–O bending vibration of anatase TiO₂. Other peaks observed at 505 cm⁻¹ (A_{1g}/ B_{1g}), and 628 cm⁻¹ (E_g) were ascribed to the stretching vibration modes of anatase TiO₂ (Benammar et al., 2020; Wang et al., 2013). It is also important to mention that unlike the XRD the Raman spectra do not show the rutile phase in the TiO₂/P25 sample and this is because Raman is slightly insensitive compared to XRD to the crystal structure of materials (Lu et al., 2013). According to several published works such as (Diego-Rucabado et al., 2020; Kruk, 2020), there are 22 active Raman lines in the



Fig. 6 (a) TEM image, (b) HRTEM image, and (c) SAED pattern of TiO₂/P25-Y₂O₃ nanocomposites.

Y₂O₃ Raman spectra (14Fg (triply degenerated), 4Eg (doubly degenerated) + 4Ag (singly degenerated)) as predicted by the theory. However, a few modes were experimentally observed which could be due to the superposition of different types of bands (Diego-Rucabado et al., 2020; Kruk, 2020). In this work, 10 characteristic lines of Y2O3 are observed at 155 cm⁻¹ (A_g + F_g), 216 cm⁻¹ (E_g), 320 cm⁻¹ (E_g), 372 cm⁻¹ (A_g + F_g), 459 cm⁻¹ (A_g + F_g), 504 cm⁻¹, 559 cm⁻¹ (A_g + F_g), 584 cm⁻¹, 634 cm⁻¹, and 728 cm⁻¹. The most intense line observed at around 372 cm^{-1} is characteristic of cubic Y₂O₃, indicating a large polarizability variation (Kumar et al., 2016). The Raman spectra of the $TiO_2/$ P25-Y₂O₃ sample show the Raman features of TiO₂ with a small peak at 365 cm⁻¹ related to Y_2O_3 . Furthermore, the characteristic peaks of TiO2 are slightly shifted and become broader, indicating the decrease of TiO2 crystallinity or/and due to the decrease of TiO₂ particle size in the nanocomposite sample. This observation is consistent with the XRD result shown in Fig. 2.

The fabricated materials were analyzed via a field-emission scanning electron microscope (FESEM) attached with energydispersive X-ray spectroscopy (EDX) to investigate the morphology of TiO₂/P25 and its dispersion on the Y₂O₃ sheet. Fig. 4 exhibits the FESEM micrographs of TiO₂/P25, Y₂O₃, and TiO₂/P25-Y₂O₃ nanocomposites fabricated by PLA. Ini-



Fig. 7 PL spectra of $TiO_2/P25$, Y_2O_3 , and $TiO_2/P25 \cdot Y_2O_3$ nanocomposites excited at 350 nm.

tially, TiO₂/P25 nanoparticles demonstrated in Fig. 4a display sizes below 40 nm and random adhesion of nanoparticles. Furthermore, it can be observed from the FESEM image that the Y₂O₃ fabricated by PLA consists of several sheets or plate-like structures attaching. The image of After TiO₂/P25-Y₂O₃ nanocomposites (Fig. 4c d) shows the loading of TiO₂/P25 nanoparticles on the surface of Y₂O₃, which confirmed the formation of TiO₂/P25-Y₂O₃ nanocomposites. Besides, Fig. 5 shows the elemental mapping of the TiO₂/P25-Y₂O₃ sample in which the elements Ti, O, and Y are mapped in green, yellow, and red colors, respectively. This mapping confirmed the laser-anchoring of TiO₂/P25 nanoparticles on Y₂O₃ sheets. Due to the unique attributes of the PLA technique, contamination-free TiO₂/P25-Y₂O₃ nanocomposites were fabricated with a straightforward procedure.

Fig. 6 (a-c) shows the TEM micrograph, High-resolution TEM, and selected area electron diffraction (SAED) pattern of $TiO_2/P25$ - Y_2O_3 nanocomposites. The dark contrast in Fig. 6a indicates the anchoring of the $TiO_2/P25$ on the surface of Y_2O_3 nanosheets. The presence of Y_2O_3 material on the $TiO_2/P25$ - Y_2O_3 nanocomposites can be seen clearly from the

high-resolution TEM image (Fig. 6b). These images confirm the non-uniform clustering of Y_2O_3 in the nanocomposites. Fig. 6c shows the SAED pattern of $TiO_2/P25-Y_2O_3$ nanocomposites. The image confirms the polycrystalline structure of the $TiO_2/P25-Y_2O_3$ nanocomposite with six diffraction rings that are consistent with the six prominent XRD diffraction peaks of the nanocomposite.

The room temperature photoluminescence (PL) spectra of the fabricated samples were performed to investigate the optical properties. PL spectrum of TiO₂/P25 (Fig. 7) excited at 350 nm displayed a wide emission band and its Gaussian fitting showed two peaks centered at 430 nm and 501 nm, which could be caused by self-trapped excitons and oxygen vacancies, respectively [14–16]. The TiO₂/P25-Y₂O₃ sample displayed emission at a wavelength of 421 nm, and the emission peak intensity is lower compared with that of TiO₂/P25 nanoparticles. This confirms that the photoexcited e/h pairs are well separated making the TiO₂/P25-Y₂O₃ suitable for many photocatalysts applications.

Fig. 8 shows the analysis of the chemical components of the TiO_2/Y_2O_3 composite by XPS investigation. A quick survey



Fig. 8 (a) XPS survey, (b) O1s, (c) Ti2p, and (d) Y3d of $TiO_2/P25-Y_2O_3$ nanocomposites.

scan revealed the presence of titanium, oxygen, carbon, and yttrium. The intensity of the O1s is the strongest, which can be attributed to the contribution of oxygen ions from the TiO₂, Y₂O₃, and adventitious contamination from oxidation and/or water. The presence of titanium and yttrium in the survey scan indicates the formation of TiO_2/Y_2O_3 composite. The carbon 1 s is the well-known adventitious carbon from atmospheric contamination. Fig. 8 (b, c, and d) show a narrow scan for O1s, Ti 2p, and Y 3d, respectively. For the O1 s peak, two prominent peaks were observed at 528.5 eV and 530.2 eV which can be attributed to the oxygen in the anatase lattice from the oxides (TiO₂) as well as oxygen defects and/or chemisorbed hydroxyls (OOH), respectively. For the Ti atoms, doublet symmetric peaks ascribed to Ti +4 were observed at approximately 457 eV and 462.5 eV which are assigned to Ti $2p_{3/2}$ and Ti $2p_{\alpha}$, respectively (Pouilleau et al., 1997). The peak separation of 5.5 eV was noted between the spin split which is consistent with reported values in the standard (Greczynski and Hultman, 2020). For the yttrium atoms, the low XPS resolution of the two spin-orbit components (Y 3d $_{5/2}$ and Y 3d $_{3/2}$ electrons) indicates the presence of multiple chemical states. The binding-energy separation of 2 eV is found between the spin-split doublets, which is consistent with the XPS standard ('X-ray Photoelectron Spectroscopy (XPS) Reference Pages: Yttrium', n.d.).

Fig. 9 shows the UV–Vis absorbance spectra of TiO₂/P25, Y_2O_3 , and TiO₂/P25- Y_2O_3 nanocomposites in the range of 200–800 nm. The absorption edge of TiO₂/P25 and Y_2O_3 occurred at approximately 279 and 283 nm, respectively. The Y_2O_3 sample shows moderate absorbance across the visible range, whereas the TiO₂/P25 sample showed absorbance in the UV region. Remarkably, TiO₂/P25- Y_2O_3 nanocomposites exhibit a very strong absorbance with the entire 250–650 nm spectrum, which is in a good agreement with the PL results. This result highlights the strong modification of the UV–VIS absorbance spectra of TiO₂ due to the presence of Y_2O_3 . Furthermore, the ionic radius of Y^{3+} is 0.088 nm, which is larger than that of Ti⁴⁺ (0.068 nm), so the Y ions should be difficult to get into the TiO₂ lattice, in the contrary, partial Ti ions may have a chance to enter the Y_2O_3 crystal lattice, thus resulting in



Fig. 9 UV–Vis absorbance spectra of TiO₂/P25, Y_2O_3 , and TiO₂/P25- Y_2O_3 nanocomposites.

crystal defects and lattice distortion. The defects levels could make new capture centers of the light generated carriers, impeding them to recombine and prolong the lifetime of electrons and holes. This new observation provides a means of increasing the absorbance of TiO_2 which can be useful in UV and visible light response applications.

5. Conclusion

This study successfully synthesized TiO₂ /Y₂O₃ nanocomposite using PLA in liquid for the first time. PLA-prepared nanostructures are unique and non-toxic because of the absence of chemical precursors or surfactants which makes nanostructures prepared via this route superior to any chemical approach. The structural, optical, and morphological properties of the nanocomposites were investigated using XRD, Raman spectroscopy, FESEM attached with EDX, photoluminescence, XPS, and UV–Vis absorbance spectra. The results obtained demonstrate that hybridizing TiO₂ with Y₂O₃ significantly modifies the light absorption characteristics within the UV–visible region. This enhancement indicates that the nanocomposite could be useful for UV and visible lightresponsive applications.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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