

## ORIGINAL ARTICLE

### King Saud University

## Arabian Journal of Chemistry

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Graphene oxide decorated bimetal (MnNi) oxide

nanoflakes used as an electrocatalyst for enhanced

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oxygen evolution reaction in alkaline media

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Received 2 August 2019; accepted 8 October 2019 Available online 16 October 2019

#### **KEYWORDS**

MnO<sub>2</sub>; NiO; Graphene oxide; Electrodeposition; Alkaline solution; Annealing; OER

Abstract Precious non-noble metals have been constantly attracting research attention in order to realize an inexpensive, extra active and more stable electrocatalysts in terms of various oxidation states and structures for their applications in oxidation (splitting) of water. In the present work graphene oxide incorporated, MnO2-NiO composite metal oxide nanoflakes were synthesized on the stainless steel substrate using efficient electrodeposition route in alkaline media and drop casting method with further annealing treatment at 400 °C for 4 h. Initially MnO<sub>2</sub>-NiO nanoflakes were deposited using different cyclic sweep rates, later graphene oxide suspension was drop casted on the MnO<sub>2</sub>-NiO nanoflakes and subsequently subjected to annealing at 200 °C for 2 h. The prepared electrode material is denoted as GO/MnO2-NiO/SS and used as an electrocatalyst for oxygen evolution. Field emission scanning electron microscopy, transmission electron microscopy, Energy dispersive electron spectroscopy and X-ray diffraction spectroscopy were used to study the crystalline nature and morphologies of the deposited films. The electrochemical properties of the electrode material were investigated using cyclic voltammetry and linear sweep voltammetry. The electrode exhibits low overpotential and small Tafel slope of 379 mV and 47.84 mVdec<sup>-1</sup> at the current density of 10 mA  $\rm cm^{-2}$  in alkaline (KOH) medium. In addition, the electrode shows a long time stability of 28800 s. Hence, the present study suggests that the GO incorporated Mn-Ni bimetal oxide

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<sup>1</sup> Future Communication

Peer review under responsibility of King Saud University.



https://doi.org/10.1016/j.arabjc.2019.10.003

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modified electrode is suitable electrode material for oxygen evolution reaction (OER), owing to its facile preparation, inexpensive, easy handling and high active nature.

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#### 1. Introduction

Global environmental problems and demand for clean energy became severe concern in recent years. The exploration, preparation and synthesis of novel energy materials with various nanostructures for high surface area have become the focus of recent research. The potentiality, clean production methods and elevated performance make hydrogen as a promising alternate candidate for the fossil fuels (Xie et al., 2103, Darabdhara et al., 2015). Hydrogen can be simply obtained from water putrefaction by using renewable energy sources such as wind and solar energies (Chu and Majumdar, 2012; Guio Morales et al., 2014). In spite of this, electrolysis of water is an enormously competent way to generate hydrogen (Chu and Majumdar, 2012; Guio Morales et al., 2014; Yan et al. 2018; Tan et al., 2017). The water catalysis involves two reactions, one is the hydrogen evolution reaction (HER) and other is the oxygen evolution reaction (OER) (Yang et al., 2017; Chi et al 2017; Wang et al., 2018; park et al., 2012]. During the catalytic reaction the catalyst can significantly reduce the energy barrier in these two reactions. Pt-based and Ir/Ru-based materials are potential mediating catalysts for water splitting. Despite of high catalytic activity of these noble metal-based catalysts, their highs cost, scarcity and poor stability limit their applications (Zou and Zhang, 2015; Lee et al., 2012; Reier et al., 2012; Antolini, 2014; Danilovic et al., 2014; Oh et al., 2015; Browne et al., 2016). In order to realize more practical and competent catalysts for extensive applications, the catalysts must be based on common materials and the methods of preparation of catalysts must be as simple as achievable. Precious non-noble metals have been continuously attracting research attention in order to understand an inexpensive, extra active and more stable electrocatalysts for their applications in oxidation (splitting) of water. Many researchers have synthesized various catalysts, based on the transition metal-based compounds including metals and metal oxides (McKone et al., 2013; Zhang et al., 2016a; Zhang et al., 2016b; Zhenhua et al., 2018; Arpita et al., 2017; Michael et al., 2015; Zhang et al., 2016c) for effectual water splitting (HER and OER).

The oxygen evolution reaction (OER) plays a significant role in evolving clean energy and storage technologies, as it produces electrons that can split the water in to hydrogen, carbon dioxide in to carbon containing fuels, and metal ions in to metals during recharge of metal-air batteries (Suntivich et al., 2011; Oh et al., 2012; Wang, 2015; Lu and Zhao, 2015). The devices associated with OER for various industrial and other applications mostly rely on low-cost, abundant and promising electrocatalysts that support slow processes involving transfer of four electrons (Morales Guio et al., 2016). Increasing the loaded mass is one of the feasible ways to attain high current at reasonable overpotential, but this method can hinder mass and charge transfer (Batchellor, 2015). Further, an additional impenetrability of catalysts observed at high OER currents results in to an exhaustive bubble release, which in turn causes unfavorable bubble-shielding effect and catalyst detachment problem (Lu and Zhao, 2015). Thus efficient electrocatalysts are required, in order to accelerate the reaction, limit the overpotential and thus improve the energy conversion competence. Various catalysts, based on the combination of metal, metal oxides/hydroxides and graphene composites, such as cobalt oxide, titanium oxide, nickel oxide, manganese oxide, Ni-Fe oxide, Ni-Mn oxide, Ni-Co oxide, Zn-Co and so on oxide based materials have been extensively developed for OER (Singh et al., 2015; Li et al., 2019; Yan et al., 2019; Wang et al., 2019; Hutchings et al., 2015; Huang et al., 2019; Yushuai et al., 2017; Wu et al., 2016; Menezes et al., 2015; Song and Hu, 2014; Rong et al., 2015; Qiao et al., 2015; Zhuang et al., 2015; Can et al., 2018; Senthil et al., 2018; Wu et al., 2019; Pei et al. 2019). Manganese is one among those metals which is an abundant element and manganese oxides (MnO<sub>2</sub>) have been previously investigated to achieve high activity for the oxygen evolution reaction (Takashima et al., 2012; Kang et al., 2017; Chen ett al., 2014; Li et al., 2015; Prasad et al., 2020). Although graphene based materials are also superior electrocatalyst (Wan et al., 2015), the better performance of MnO<sub>2</sub>-NiO remains unchallenged by any pure graphene materials. Yang et al. (2012) used MnO<sub>2</sub>-graphene nanosheets as the air cathode in lithium-air cells and obtained a discharge capacity much higher than pure graphene nanosheets (GNS) catalyst. For further improvement of the properties OER, significant efforts have to be put forward by preparing metal oxides and graphene based composite materials.

However few reports were found in the literature, on the combination of manganese, nickel oxide and graphene oxide composite materials previously. Hwang et al. (2013) reported on preparation of graphene-NiO-MnO<sub>2</sub> nanocomposite. The catalyst was formed by a simple chemical precipitation method using a chelating agent and Ni and Mn hydroxides on graphene for supercapacitors applications (Wan et al., 2015). Rusi et al. (2016) investigated on the effects of electrodeposition mode and deposition cycle of NiO-MnO<sub>2</sub> composite electrodes on the electrochemical performance for high energy density supercapacitors. Lee et al. (2014) prepared manganese-nickel oxide films on a graphite sheet using electro-deposition technique for electrochemical capacitor applications. Tran et al. (2015) synthesized nanoflake manganese oxide and nickel-manganese oxide using electrodeposition for electrochemical capacitors applications. Ye et al. (2017) reported on various metal ions doped-MnO<sub>2</sub> ultrathin nanosheets for catalysis applications. All the above methods either involve a separate synthesis of each material, followed by combining of all the materials once or layer by layer synthesis in multistep process by electro-deposition method. However it is found that there is a difficulty in achieving homogeneous nanostructures over the substrates. In case of our method the electrodeposition of bimetal hydroxides deposited at a single

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step in alkaline media with homogenous nanostructure and it is significant, very simple, selective and sensitive compared to the literature methods reported in the literature.

In the present work we developed a facile method for preparing the electrode using a composite of bimetallic oxides (Mn-Ni) and graphene oxide by employing combined simple electrodeposition and drop casting methods in alkaline solution. Subsequent annealing for a short period led to form a homogenous nanostructure. In this method deposition cycles, annealing time and annealing temperatures play a significant role in the formation of homogeneous nanostructure. To the best of our knowledge we didn't find any report on the method which employs the combination of the materials particularly for oxygen evolution reaction.

#### 2. Experimental

#### 2.1. Materials and reagents

The stainless steel foil with 99% purity (with measurements  $15 \times 35 \times 1$  mm) was purchased from Good fellow, China. All chemicals, manganese acetate tetra hydrate  $(Mn(CH_3COO)_2 \cdot 4H_2O),$  nickel acetate tetra hydrate (Ni(CH<sub>3</sub>COO)<sub>2</sub>·4H<sub>2</sub>O), sodium sulphate (Na<sub>2</sub>SO<sub>4</sub>), sodium hydroxide (NaOH), sulfuric acid (H<sub>2</sub>SO<sub>4</sub>) and graphene oxide (GO) which are standard analytically reagent graded, were purchased from Sino pharmchemical reagent company and used as usual without additional purification. Deionized water (DI) was used in all aqueous solutions.

#### 2.2. Instrumentation for characterization

All electrochemical experiments such as cyclic voltammetry, linear sweep voltammetry, amperometry and electrochemical impedance spectroscopy (EIS) were carried out using a CHI 660E (CH Instrument Company of Shanghai, China) electrochemical work station (potentiostat/galvanostat). A three electrode system which contains GO/MnO2-NiO modified electrode, Pt wire and Ag/AgCl, SCE as working electrode; counter electrode and reference electrode respectively is used in 0.1 M KOH solution. For EIS, the applied frequencies are from 0.1 Hz to 10 kHz. X-ray diffraction studies were carried out (model XRD, Bruker AXS, Germany) using the Cu Ka radiation ( $\lambda = 1.5418$  Å) and field-emission scanning electron microscopy (FE-SEM, FEI Nov. 450 Nano, USA) were used to study the morphology and nanostructure of the prepared electrodes. Further Transmission electron microscopy (TEM), (JEM-2100UHR STEM/EDS, JEOL, Japan) was used for characterization of the electrodes.

## 2.3. Preparation of graphene oxide incorporated MnO<sub>2</sub>-NiO nanoflakes

The stainless steel sheet  $(35 \times 15 \times 1 \text{ mm})$  of model 316 L with 99% purity was used as substrate. Initially the stain steel substrate was mechanically polished using emery paper of different grades 500, 1000 and 1500, and an aluminium slurry as well, for attaining the mirror like surface. Subsequently the as polished substrate was cleaned ultrasonically in deionized water and ethanol for 10 min, and then dried in the stream

of pure N<sub>2</sub>. The deposition of electrolyte involve preparation of the mixture of 0.02 M Manganese (II) Acetate tetrahydrate (Mn(CH<sub>3</sub>COO)<sub>2</sub>·4H<sub>2</sub>O), 0.05 M Nickel acetate tetrahydrate (Ni(CH<sub>3</sub>COO)<sub>2</sub>·4H<sub>2</sub>O), and 0.5 M sulphuric acid (H<sub>2</sub>SO<sub>4</sub>) or 1.0 M Na<sub>2</sub>SO<sub>4</sub> in a 250 mL beaker which is stirred magnetically for 30 min, and then obtained a homogeneous solution. Further, adequate electrolyte solution was transferred in to the electrochemical cell containing a three electrode system that employs electrodeposition method (cyclic sweep). The applied potentials at different scan rates (5-50 mV/s) are in the range of 0-1.4 V. A set of different cycles (6, 10, 20 cycles) for various alkaline and acidic solutions [for acidic solution see supplementary information (SI)], such as Na<sub>2</sub>SO<sub>4</sub> and H<sub>2</sub>SO<sub>4</sub> are employed. The obtained nano composite were annealed at various temperature and time intervals from 300 to 400 °C for 1.0 h to 6.0 h and finally got the uniformed nanoflakes. Here the formation of flakes occurred may be the oxidation state of Mn increases. Since it is improbable that two electrons are transferred in a single step, the formation of MnO<sub>2</sub> from  $Mn^{2+}$  occurs concomitantly with the oxidation of water molecules. Similarly nickel oxide makes simultaneously with MnO<sub>2</sub> use of nickel hydroxide precipitation. Nickel hydroxide is formed on the electrode, finally, the electrodeposition of nickel/manganese oxides in an electrolyte containing  $Mn^{2+}/$ Ni<sup>2+</sup> ions (Lee et al., 2014). Based on the cyclic voltammetric response, the applied cyclic voltammetric cycles of 10 CV cycles at scan rate 20 mV/s (20/10) in alkaline solution were selected as optimum conditions and the obtained electrode was used for further analysis. Under these conditions well defined MnO<sub>2</sub>-NiO nano flakes were obtained. Finally different amount of graphene oxide (10-50 µL) was drop casted on the as prepared MnO2-NiO nano flakes, dried in air 5-6 h, then annealed at 200 °C for 2 h. During the annealing process the graphene oxide formed into a homogeneous layer on MnO<sub>2</sub>-NiO nano flakes and the nano composite loaded on the substrate around 0.1 mg. The final obtained electrode



Scheme 1 The schematic representation of the electrode preparation at optimum conditions.

denoted as  $GO/MnO_2$ -NiO (10 GO, 20 GO, 30 GO and 50 GO), and used for further analysis. The schematic illustration of electrode preparation was depicted in Scheme 1.

#### 3. Results and discussions

#### 3.1. The electrochemical performance of the electrode

The bimetal oxide based nanostructure was deposited on the steel substrate by electro-deposition method (cyclic sweep). The bimetal hydroxide (OH) based nanostructure was formed on steel substrates by cyclic voltammetric sweeping between 0 V and 1.4 V at various scan rates  $20 \text{ mV s}^{-1}$ ,  $50 \text{ mV s}^{-1}$  for different cycle segments after that annealed at  $400^{\circ}$ C for 4 h. Fig. 1, shows the cyclic voltamogrammes obtained at 50 mV s<sup>-1</sup> for 6 segments (50/6) (A), 50 mV s<sup>-1</sup> 10 segments (50/10) (B),  $20 \text{ mV s}^{-1}$  for 20 segments (20/20) (C),  $20 \text{ mV s}^{-1}$  for 10 segments (20/10) (D). The graphene oxide was drop casted on the as prepared MnO<sub>2</sub>-NiO nanoflakes and subsequently annealed at 200°C for 2 h and GO layer was formed and exhibited various colours. The photographs of the as prepared electrodes are shown in Fig. S2. The prepared

nanoflakes exhibited an improved electrochemical activity and surface area. The well-defined peeks were obtained at 20 mV s<sup>-1</sup> for 10 (20/10) segments [Fig. 1D], and hence was selected as optimised sample and used for further analysis. The expected mechanism of electrode reaction based on bimetal oxide employs cyclic sweep mode can be described as follows (Scheme 2). Initially during the deposition process when the potential is applied water, molecule reduced at working electrode and produced hydroxyl ions. The as produced  $OH^$ ions play a significant role in the growth of metal oxides over the steel substrate. It may be the instantaneous nucleation process. Then, a progressive nucleation process takes place, in which the flake structure grows. As the number of deposition segments (cycles) increases, the growth rate of the progressive nucleation also increases, resulting the nano flakes when main-







Fig. 1 Cyclic voltamogrammes of the preparation electrode (A) deposition at 50 mV/s, 6 segments (50/6), (B) deposition at 50 mV/s, 10 segments (50/10), (C) deposition at 20 mV/s, 20 segments (20/20), (D) deposition at 20 mV/s 10 segments (20/10) in 1.0 M Na<sub>2</sub>SO4 solution. (Optimum is 'D').

tained the scan rates at 20 mV at 10 segments. The formation of  $Mn^+$  ion and  $Ni^+$  ion occurs, where the hydroxide (OH<sup>-</sup>) ions are localized. The metal hydroxide nanostructure was formed between  $Mn^+/Ni^+$  cations and OH<sup>-</sup> ions. Finally bimetal oxide nanoflakes were obtained on the substrate during the annealing process (Rusi et al., 2014).

#### 3.2. Morphological studies

The surface morphology of GO/MnO<sub>2</sub>-NiO/SS electrodes was studied by field emission scanning electron microscopy. The SEM images for electrodes employed various cyclic voltammetric scan rates and cycles, were shown in Fig. S3. Based on the observed, well defined cyclic voltamogrammes (Fig. S4) and well organised nanostructure [Fig. 2 (A, B different magnifications] of the sample obtained at scan rate 20 mV/ s for 10 segments and subsequent annealing at 400°C for 4 h, the sample is denoted as (20/10), it was selected as the suitable and optimum sample and used for further application. Fig. 2 clearly shows that the nanoflakes of bimetal oxide (MnO<sub>2</sub>-NiO) nano structure formed over the steel substrate. Further,

Fig. S3 (E) and Fig. S3 (F) present the SEM images of MnO<sub>2</sub>-NiO nanoflakes incorporated with 10  $\mu$ L and 50  $\mu$ L of graphene oxide respectively. The graphene oxide layer was clearly observed on MnO<sub>2</sub>-NiO nanoflakes. Based on the quality appearance of nanostructure the nano flakes formed by 30  $\mu$ L of GO was selected as optimum sample, [Fig. 2(C)]. In addition Energy-dispersive X-ray spectroscopy (EDX) was also carried out for the electrodes; The EDX recorded at the portion, marked by pink colored square of Fig. 2(A), indicates the presence of Mn, Ni, and O elements and carbon, originated from graphene oxide in the sample, the corresponding EDX spectra shown in Fig. S5 It can be confirmed that graphene oxide and MnO<sub>2</sub>-NiO nanoflakes were well formed on the steel substrate.

X-Ray diffraction studies were used to examine the crystalline nature and structural details of nano structure of bimetal oxide (Mn-Ni). The obtained XRD patterns of bimetal oxide (Mn-Ni) based nanoflakes are shown in Fig. 3. The XRD peaks observed were attributed to MnO<sub>2</sub>-NiO nanoflakes and can be indexed to MnO<sub>2</sub> [JCPDS card no. 44-0141] and NiO [JCPDS card no. 78-0643]. The sharp and



Fig. 2 SEM images of the electrode  $MnO_2$ -NiO flakes (A,B different magnifications) at 20/10, and graphene oxide decorated MnO2-NiO nanoflakes at 30 GO (C).



Fig. 3 XRD patterns of the as prepared electrode  $30 \,\mu\text{L}$  GO/MnO<sub>2</sub>-NiO (30GO) electrode, annealing at  $200^{\circ}$  C for 2 h and, (Inset enlarged view of XRD patterns).

intense diffraction peaks of MnO2-NiO flakes confirm that the samples possess good crystalline nature. A broad hump observed at approximately  $2\theta \sim 12^{\circ}$  which can be ascribed to the amorphous phase of the nanostructure. The semi crystalline and amorphous phases have their origin in the employed cyclic sweeps, scan rate, deposition time, and annealing temperature. The peaks found at angles  $2\theta \sim 28.89^{\circ}$  and  $2\theta \sim 37.5^{\circ}$ ,  $2\theta \sim 41.9^{\circ}$ ,  $2\theta \sim 56.3^{\circ}$ , and  $2\theta \sim 65.1^{\circ}$  which correspond to the (310), (211), (301), (600) and (002) respectively, are attributed to the  $MnO_2$ and the peaks observed at around  $2\theta \sim 43.8^{\circ}$  that correspond (200) diffraction planes, is related to NiO (Hawang et al., 2013; Rusi et al., 2014) Some electrodes (prepared at high scan rates and more cycles) had shown low intensity XRD peaks, which explains that the MnO2-NiO nanostructures possess poor crystallinity, this is in contrast with the nanoflakes having a superior specific area (Rusi et al., 2016). Furthermore a wellresolved XRD peak was found at around  $2\theta \sim 12.0^{\circ}$ , which corresponds to the graphene oxide incorporated in to prepared electrode, the enlarged view of (GO/MnO2-NiO) 30 GO XRD patterns are shown in Fig. 3. The present results were compared with existing reports and found that good agreement with those reported methods (Rusi et al., 2016; Lee et al., 2014; Rusi et al., 2014).

The TEM studies were carried out to investigate the microstructural details of the as prepared  $MnO_2$ -NiO nano structure based electrodes. The monitored TEM images are shown in Fig. 4, where Fig. 4 (a) depicts a cluster of 30 GO composite which clearly reveals rumple like structure [inset] which is attributed to the nanoflake morphology. The graphene oxide layers [marked using red coloured arrows in Fig. 4(a)], are found to be well combined with the  $MnO_2$ -NiO composite, indicating that the graphene oxide, well composited with the nanoflake  $MnO_2$ -NiO composite. The distinc-

tive composite structure and homogeneous nano flake structure play a significant desired functional role for electrochemical application as an active material with a quick diffusion rate through the redox phase. The selected area electron diffraction (SAED) pattern of several thin sheets of the 30 GO (GO/MnO<sub>2</sub>-NiO) [Fig. 4(b)] displays two sets of intense and bright spots forming clear rings correspond to the both MnO<sub>2</sub> and NiO nano composite structure, signifying the related crystallinity. The observed intense SAED ring patterns are indexed to (1 1 0), (1 0 1) and (2 1 1), planes of MnO<sub>2</sub> for dspacing, ~0.311 nm, ~0.240 nm and ~0.162 nm respectively according to the JCPDF card No: 81-2261. Additional reflections from (1 1 1) and (2 0 0) planes correspond to the NiO for d-spacing ~0.241 nm and ~0.208 nm respectively are found to be in accordance with the JCPDF card No: 78-0643. These observations from SAED patterns are in good agreement with that of XRD studies. In addition, HRTEM image [Fig.(c)] depicts highly crystalline interconnected lattice portions and amorphous layer located at the edges of crystalline portions. The deeper insight in to the micro structure of the lattice is obtained by analysing the high-resolution TEM (HRTEM) lattice image [Fig.(d)]. The FFT images [insets of Fig. 4(d)] represent two distinguished lattice portions that correspond to the MnO<sub>2</sub> and NiO. The lattice fringes of MnO<sub>2</sub> crystalline structure were viewed along (111) and indexed for atomic planes(110), (011), and (101), with lattice spacing  $\sim 0.311$  nm. ~0.240 nm and ~0.240 nm respectively. The lattice planes correspond to NiO, (111), (111) and (200) were well resolved when viewed along the zone axis (011) which are belonging to the inter planar spacings of ~0.241 nm, ~0.241 nm and  $\sim 0.208$  nm respectively (Rusi et al., 2016; Lee et al., 2014; Rusi et al., 2014). These observations ensure that the nano crystalline structures of MnO<sub>2</sub> and NiO co-exist in to a



Fig. 4 TEM images of  $MnO_2$ -NiO nanoflakes with GO (a), SAED patterns of  $MnO_2$ -NiO nano flakes (b), HRTEM image of 30 GO (c), and HRTEM image of 30 GO with FFT  $MnO_2$ -NiO (d).

composite form. In addition, it is clearly found that the  $MnO_2$ -NiO composite is well combined with GO which is shown using two parallel lines at the edges of the composite nano structure. These observations are in consistent with that of FESEM, XRD and TEM studies.

Electrochemical impedance spectroscopy (EIS) was extensively used for studying the electron transfer between the electrolyte and the electrode surface employed. The impedance plot is usually characterized by a combination of semicircular arch of high frequency region and a straight line portion of low frequency range. The earlier corresponds to the charge (electron) transfer resistance (Rct) and can be obtained from the diameter of the arch, while the later, is associated to the electron diffusion process. The charge transfer resistance (Rct) influences the conductivity of the electrode and thus electron transfer in to the electrochemical reaction process. Fig. 5 depicts the Nyquist plots monitored for different electrodes under the applied frequency ranging from 0.01 Hz to 100 kHz. It can be clearly seen from the figure that the Nyquist plot of MnO<sub>2</sub>-NiO electrode exhibit the largest semicircle with some portion as straight line. The diameter of the semi-circle is measured on the X-axis as Z'Ω, which indicates that the electrode has very high charge transfer resistance and a poor electron diffusion, during reaction process. This indicates poorer electrochemical activity and lower conductivity of MnO2-



Fig. 5 Nyquist plots from EIS for all referred electrodes [(a)  $MnO_2$ -NiO/SS, (b) 50 µL GO-MnO\_2-NiO/SS without annealing, (c) 30 µL GO-MnO\_2-NiO/SS, (d) 20 µL GO-MnO\_2-NiO/SS (e) 10 µL GO-MnO\_2-NiO/SS] in 0.1 M KOH solution applied frequency between 100 kHz and 0.01 Hz.

NiO electrode. The measured values of Rct for MnO<sub>2</sub>-NiO flakes treated with different amounts of graphene oxide are ranging from 200  $\Omega$  to 600  $\Omega$  as shown in the Fig. 5(a,b). The Rct of the MnO<sub>2</sub>-NiO electrodes gradually decreases with the amount of graphene oxide decreased from 50  $\mu$ L to 10  $\mu$ L (followed the order c < e < d < b < a). It is found that the 30µLGO/MnO<sub>2</sub>-NiO (30 GO) exhibits the lowest charge transfer resistance, signifying that the electrolyte sample possess a good electron diffusion mechanism, which is possibly due to the interconnected structure of obtained nano flakes and GO, and is in good agreement with the observed SEM and TEM images. The above results confirm that the 30 GO electrode has superior electrochemical activity than the remaining electrodes, the order (a < b < d < e < c) is indicating that the electrode exhibit a synergistic effect of the constituent nanomaterials.

3.3. Electrochemical performance of the electrode towards oxygen evolution reaction (OER)

Linear sweep voltammetry was used in order to demonstrate the elastrochemical performance of the electrodes, the OER behavior of MnO<sub>2</sub>-NiO flakes and GO/MnO<sub>2</sub>-NiO are examined through in 0.1 M KOH at a sweep rate of 10 mV/s, and

 $\begin{array}{l} Mn/Ni2^{+} + 3OH^{-} \leftrightarrow Mn/NiOOH + H_{2}O + e^{-} \\ Mn/Ni(OOH) + OH^{-} \leftrightarrow Mn/NiOOH) + e^{-} \\ Mn/NiO(OH)_{2} + OH^{-} \leftrightarrow Mn/Ni(OO_{2}) + 2H_{2}O + 2e^{-} \\ Mn/Ni(OO_{2}) + OH^{-} \rightarrow Mn/NiOOH + O_{2} + e^{-} \\ Overall OER: 4OH^{-} \leftrightarrow O_{2} + 2H_{2}O + 4e^{-} \end{array}$ 



Scheme 3

**Fig. 6** Linear sweep voltammetric curves of various electrodes [A] (a) SS, (b)  $MnO_2$ -NiO, (c)  $MnO_2$ -NiO with annealing, (d) 10 GO, (e) 20 GO, (f) 50 GO, (g) 30 GO (inset enlarged view of LSV curves, [B] Tafel plates for various electrodes [(a) SS, (b)  $MnO_2$ -NiO, (c)  $MnO_2$ -NiO with annealing, (d) 10 GO, (e) 20 GO, (f) 50 GO, (g) 30 GO], [C] Durability of the 30 GO electrode for 100 cycles. [D] Durability of the various electrodes  $MnO_2$ -NiO and 30 GO using chronopotentiometry at j: 10 mV/cm<sup>2</sup>.

Table 1 Comparison of OEK activity on with and Al related electrodes/catalysis.			
Electrode/Catalyst	Overpotential (V)	Tafel slope (mV/Dec)	Ref
30 µL GO/MnO <sub>2</sub> -NiO	0.379	47.84	Present
Mn-Ru Oxides	$0.259 \pm 0.016$	$48 \pm 2$	Browne et al. (2016)
MnO <sub>X</sub>	0.60	120 to 70	Michael et al. (2015)
NiO <sub>x</sub> , MnO <sub>x</sub> , and FeO <sub>x</sub>	0.24-0.27	32–36	Morales Guio et al. (2016)
NiO/Ni nanosheets	0.26	51	Yushuai et al., 2017
CoMn LDH	0.35	43	Song and Hu (2014)
Ni-Fe-OH	0.316	53	Wu et al. (2019)
MnO <sub>X</sub> bi layers	0.45	-	Kang et al. (2017)
Gr-MnO <sub>2</sub>	0.20	-	Chen et al. (2014)
H-TiO <sub>2</sub> /MnOx NWs	0.524	-	Li et al. (2015)

Table 1 Comparison of OER activity on Mn and Ni related electrodes/catalysts

the results are shown in Fig. 6. It is found that the potentials vs RHE of oxygen evolution reaction are 1.768 V, 1.689 V, 1.658 V, 1.64 V, 1.62 V and 1.609 V and corresponding to overpotential are 0.539 V, 0.458 V, 0.43 V, 0.418 V, 0.398 and 0.379 V for electrodes based on the MnO<sub>2</sub>-NiO, MnO<sub>2</sub>-NiO annealing, 10 GO, 20 GO, 50 GO and 30 GO, respectively.

The lowest onset potential is exhibited by 30 GO based electrode indicating that it is highly active anode materials towards OER application. Hence a fast OER activity can easily be achieved by the 30 GO electrode. The possible reaction mechanism was shown in Scheme 3. Further the current density values measured from the polarization curves obtained under the applied potentials ranging from 0.6 V to 0.8 V for the prepared electrodes based on different amounts of graphene oxide at  $10 \text{ mA cm}^{-2}$  current density. It can be seen from the Fig. 6A, that the OER activity was increased up to  $30 \,\mu\text{L}$  of graphene oxide and then decreased with the concentration of GO. The substrate SS (a), MnO<sub>2</sub>-NiO (b), annealed MnO<sub>2</sub>-NiO (c) showed poor activity of the OER and the MnO<sub>2</sub>-NiO flakes treated with graphene oxide (d-g), exhibited a higher activity suggests that the electrocatalyst is a potential catalyst for achieving significant activity. The enhanced OER activity and current density can be related to the flake like nanostructure of MnO<sub>2</sub>-NiO-GO, increased electrode surface area and also the synergistic effect between the metal oxides. The photography was taken while water oxidation process shown in Fig. S6, we can see that the bubbles were clearly formed on the working electrode. It's evident that the as prepared electrode exhibits good OER catalytic activity.

The Tafel slope of the as prepared electrodes serves a key indicator to assess the performance of oxygen evolution. The Tafel slopes measured from the linear portion of the curves [Fig. 6B] of the range potentials vs RHE around from 1.6 V to 1.8 V for various electrodes (different cyclic sweeps, scan rates and graphene oxide) are given in Table 1 which indicates good OER activity. From the Table 1 it can be observed that 30 µL GO/MnO<sub>2</sub>-NiO (30 GO) electrode displays the lower Tafel slope of  $47.84 \text{ mV} \text{ dec}^{-1}$ , which is similar to that of bench mark electro catalysts (Browne et al., 2016; Chen et al., 2014; Li et al., 2015). The obtained results suggest that the electrode 30 µL GO/MnO<sub>2</sub>-NiO exhibits a good catalytic performance of oxygen evolution. The electrodes especially graphene oxide deposited MnO<sub>2</sub>-NiO flakes (10-50 µL GO/ MnO<sub>2</sub>-NiO) exhibit Tafel slopes lying between 58 and 49 mV/dec. The lower Tafel slopes reveal that the graphene oxide layer was deposited well with metal oxides nanoflakes and played significant role in OER behaviour.

Furthermore the stability of the electrodes was examined by linear sweep voltammetry and chronoamperometry. As shown in Fig. 6C, the comparative LSV curves up to 100 cycles have no big change in the current density and OER activity. Furthermore according to the stability study using chronoamperometry both the electrodes  $MnO_2$ -NiO and 30 µL GO/MnO\_2-NiO (30 GO) exhibit good stability up to 28,800 sec, no dispersion from the straight behavior or fall down in the curve is found, as seen from the Fig. 6D. The results clearly prove that when the graphene oxide decorated, the OER potential was decreased it shows that as prepared electrodes exhibit good OER stability.

#### 4. Conclusions

The bimetal oxides (MnO<sub>2</sub>-NiO) and graphene oxide mixed composite electrodes GO/MnO2-NiO were synthesized and their anodic water oxidation activities in an aqueous alkaline solution were studied. The main parameters such as mixture of (MnO<sub>2</sub>-NiO-GO) oxide composition, deposition cycles, scan rate, annealing temperature, and annealing time influenced the development of the nanostructure and OER activity. The combination of these mixed oxides shows superior activity for oxygen evolution at a low overpotential. The electro catalyst contained GO and employed Mn-Ni precursor solutions deposited at various scan rates (10-50 mV/s for 6-20 cycles, show good OER activity regarding overpotential and Tafel slopes, present results are comparable to that of previous reports and hence the present novel composite electrode material is feasible for electrocatalytic reaction. The present method significantly contributes to the development of ongoing research by employing the combination of transition metal oxides for OER catalysts. The enhanced OER activity was attributed to the changing electronic state of the metal oxides and the crystallinity of the nanostructure was influenced by the annealing temperature and time. The present method of synthesis of new combination of the materials is less expensive, simple and feasible for implementation in comparison with that of the expensive pure Ru and Ir materials. Further, when compared with the platinum group oxides combined with the first row transition metal oxides, the present method facilitates high OER activity, and hence the composite is a promising alternative electrocatalyst for OER application.

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#### Acknowledgment

This work was supported by the National Natural Science Foundation of China under the Grant Nos. 51661033, 51361028, 51161025 and 31660538.

#### **Declaration of Competing Interest**

There are no conflicts to declare.

#### Appendix A. Supplementary material

Supplementary data to this article can be found online at https://doi.org/10.1016/j.arabjc.2019.10.003.

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